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12 Jun 2018

High Efficiency, High Performance Metal-Organic Framework (MOF) Membranes in Hollow Fibers and Tubular Modules

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Recommended Citation

S. Nair et al., "High Efficiency, High Performance Metal-Organic Framework (MOF) Membranes in Hollow Fibers and Tubular Modules," U.S. Patents, Jun 2018.

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(12) **United States Patent** (10) Patent No.: US 9,994,501 B2
Nair et al. (45) Date of Patent: $*$ Jun. 12, 2018

(54) HIGH EFFICIENCY, HIGH PERFORMANCE
METAL-ORGANIC FRAMEWORK (MOF) MEMBRANES IN HOLLOW FIBERS AND TUBULAR MODULES

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- (*) Notice: Subject to any disclaimer, the term of this $\frac{CN}{CN}$ patent is extended or adjusted under 35 U.S.C. 154(b) by 50 days.

This patent is subject to a terminal dis

- (21) Appl. No.: 15/061,964
- (22) Filed: Mar. 4, 2016

(65) **Prior Publication Data**

US 2016/0184798 A1 Jun. 30, 2016

Related U.S. Application Data

- (63) Continuation-in-part of application No. $14/231,871$, filed on Apr. 1, 2014, now Pat. No. 9,687,791.
(Continued)
- (51) Int. Cl.
 $C07C$ 7/144 **B01D 63/06** (2006.01)
 (2006.01)

(Continued)

(52) U . S . CI . CPC C07C 7 / 144 (2013 . 01) ; B01D 63 / 063 $(2013.01);$ **B01D 67/0051** (2013.01); (Continued)

(45) Date of Patent: *Jun. 12, 2018

(58) Field of Classification Search CPC B01D 67/0051; B01D 53/228; B01D 2053/224; B01D 63/063; Y10T 29/49345; CO7C 7/144; B01J 19/2475
See application file for complete search history.

OTHER PUBLICATIONS

Ameloot, et al., Interfacial synthesis of hollow metal-organic framework capsules demonstrating selective permeability, Nature Chemistry 3 (May 2011) 382-387.

(Continued)

Primary Examiner — Lessanework Seifu

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(57) ABSTRACT

A reactor cell for measuring gas and liquid permeation is disclosed. A hollow fiber is supported by and sealed into a first hole and a second hole of the reactor module. The first
and second ends of the hollow fiber are sealed with a sealing solution. Methods for making and using the reactor cell are also disclosed. As made and used, the reactor cell further comprises a molecular sieving membrane that is uniform and free of defects grown on an inner bore surface of the hollow fiber.

30 Claims, 43 Drawing Sheets $(27 of 43 Drawing Sheet(s) File d in Color)$

- (60) Provisional application No. $61/913,592$, filed on Dec.
9, 2013, provisional application No. $61/820,489$, filed on May 7, 2013.
- (51) Int. Cl.
 $B01D 67/00$ **B01D 69/08** (2006.01) (2006.01)
- (52) U.S. Cl.
CPC B01D 69/08 (2013.01); Y02P 20/582 (2015.11); *Y10T* 29/49345 (2015.01) OTHER PUBLICATIONS

U.S. PATENT DOCUMENTS

Related U.S. Application Data FOREIGN PATENT DOCUMENTS

Brown, et al., Continuous polycrystalline zeolitic imidazolate
framework-90 membranes on polymeric hollow fibers, Angew.

Framework - 90 membranes on polymeric holds and polymeric holds and processes for a sustainable industrial growth, RSC Adv. 3 (2013) 5694-5740.

Bux, et al., Oriented zeolitic imidazolate framework-8 membrane with sharp H_s/C_3H_8 molecular sieve separation, Chem. Mater. 23 (2011) 2262-2269.

Cao, et al., Growth of uniformly oriented silica MFI and BEA zeolite films on substrates, Science 334 (Dec. 16, 2001) 1533-1538.

Choi, et al., Grain boundary defect elimination in a zeolite membrane by rapid thermal processing, Science 325 (Jul. 31, 2009) 590-593.

Gascon, et al., *Accelerated synthesis of all-silica DD3R and its performance in the separation of propylene/propane mixtures*,

Microporous & Mesoporous Materials 115 (2008) 585-593.
Gascon, et al., *Practical approach to zeolitic membranes and coatings: State of the art, opportunities, barriers, and future per-*

spectives, Chem. Mater. 24 (2012) 2829-2844.
Huang, et al., *Steam-stable zeolitic imidazolate framework ZIF-90* membrane with hydrogen selectivity through covalent functionaliza-
tion, J. Am. Chem. Soc. 132(44) (2010) 15562-15564.

Jang, et al., *Modified mesoporous silica gas separation membranes* on polymeric hollow fibers, Chem. Mater. 23 (2011) 3025-3028.

Kwon, et al., Highly propylene-selective supported zeolite-
imidazolate framework (ZIF-8) membranes synthesized by rapid microwave-assisted seeding and secondary growth, Chem. Commun. 49 (2013) 3854-3856.

Li, et al., Zeolitic imidazolate frameworks for kinetic separation of propane and propene, J. Am. Chem. Soc. 131(30) (2009) 10368-10369.

Pan, et al., Effective separation of propylene/propane binary mixtures by ZIF-8 membranes, J. Membrane Science 390-391 (2012) 93-98.

Pan, et al., Sharp separation of C2/C3 hydrocarbon mixtures by zeolitic imidazolate framework-8 (ZIF-8) membranes synthesized in aqueous solutions, Chem. Commun. 47(37) (Oct. 7, 2011) 10275-10277.

Pan, et al., Synthesis of ceramic hollow fiber supported zeolitic
imidazolate framework-8 (ZIF-8) membranes with high hydrogen

permeability, J. Membrane Science 421-422 (2012) 292-298. Park, et al., *Exceptional chemical and thermal stability of zeolitic* imidazolate frameworks, PNAS 103(27) (Jul. 5, 2006) 10186-10191.

Pera Titus, et al., Preparation of inner-side tubular zeolite NaA membranes in a semi-continuous synthesis system, J. Membrane
Science 278 (2006) 401-409.

Shah, et al., Current status of metal-organic framework membranes for gas separations: Promises and challenges, Ind. Eng. Chem. Res.

for gas separations 12179-2199.
Thompson, et al., *Hybrid zeolitic imidazolate frameworks: Controlling framework porosity and functionality by mixed-linker synthesis*,

Chem. Mater. 24 (2012) 1930-1936.
Tsapatsis, *Toward high-throughput zeolite membranes*, Science 334 (Nov. 11, 2011) 767-768.

Varoon, et al., Dispersible exfoliated zeolite nanosheets and their application as a selective membrane, Science 334 (Oct. 7, 2011) 72-75.

PCT Aug. 22, 2014 International Search Report and Written Opinion issued in International Application PCT/US14/133169.

OTHER PUBLICATIONS

Aug. 3, 2012 Office Action *Non-Final Rejection mailed in U.S.* Appl. No. 12/971,132, filed Dec. 17, 2010. Dec. 18, 2012 filed in U.S. Appl. No. 12/971,132, filed Dec. 17, 2010.

Jan. 28, 2013 Office Action/Final Rejection mailed in U.S. Appl.
No. 12/971,132, filed Dec. 17, 2010.
Jun. 24, 2013 Response to Final Office Action dated Jan. 28, 2013
with Declaration filed in U.S. Appl. No. 12/971,132, f

2010.

Aug. 14, 2013 Office Action/Non Final Rejection mailed in U.S.

Appl. No. 12/971,132, filed Dec. 17, 2010.

Feb. 11, 2014 Response to Office Action dated Aug. 14, 2013 with

Declaration filed in U.S. Appl. No. 12/97

Mar. 3, 2014 Notice of Allowance/Allowability with Examiner's Amendment and Statement of Reasons for Allowance mailed in U.S. Appl. No. 12/971,132, filed Dec. 17, 2010.

Oct. 31, 2013 Office Action/Non-Final Rejection mailed in U.S.
Appl. No. 13/396,411, filed Feb. 14, 2012.
Jan. 17, 2014 Response to Office Action dated Oct. 31, 2013 filed
in U.S. Appl. No. 13/396,411, filed Feb. 14, 2012.

Apr. 9, 2014 Final Office Action mailed in U.S. Appl. No.
13/396,411, filed Feb. 14, 2012.
Jun. 3, 2014 Response to Final Office Action filed in U.S. Appl. No.
13/396,411, filed Feb. 14, 2012.
Jun. 11, 2014 Notice of Allow

Amendment/Comment mailed in U.S. Appl. No. 13/396,411, filed Feb. 14, 2012.

Dec. 19, 2013 Notice of Allowance/Allowability with Examiner's Amendment/Comment mailed in U.S. Appl. No. 13/399,645, filed Feb. 17, 2012.

Apr. 19, 2013 Office Action Non-Final mailed in U.S. Appl. No. 13/209,957, filed Aug. 15, 2011.
Jun. 14, 2013 Response to Office Action dated Apr. 19, 2013 in U.S.
Appl. No. 13/209,957, filed Aug. 15, 2011.
Jul. 11, 2013

Moloy, et al., *High-silica zeolites: A relationship between energetics* and internal surface areas, Microporous & Mesoporous Materials 54 (2002) 1-13.

Nair, Nanoscopic metal oxide objects via controlled creation and
rearrangement of amorphous nanoparticles, presented Dec. 11,

2007, School of Chemical & Biomolecular Engineering Georgia
Institute of Technology, Atlanta, GA 30332-0100.
Skoulidas, et al., *Self-diffusion and transport diffusion of light gases*
in metal-organic framework materials a Chen, et al., *Interwoven metal-organic framework on a periodic minimal surface with extra-large pores*, Science 291 (2001) 1021-1023.

Eddaoudi, et al., Design and synthesis of metal-carboxylate frameworks with permanent microporosity, Topics in Catalysis 9 (1999) 105-111.

Foster, et al., *A geometric solution to the largest free-sphere problem in zeolite frameworks*, Microporous & Mesoporous Materials 90(1-3) (2006) 32-38.

Hoshen, et al., Percolation and cluster distribution. I. Cluster multiple labeling technique and critical concentration algorithm,

Physical Review B 14(8) (1976) 3438.
Keskin, et al., *Efficient methods for screening of metal organic framework membranes/or gas separations using atomically detailed models*, Langmuir 25(19) (2009) 11786-11795.

Li, et al., *Design and synthesis of an exceptionally stable and highly porous metal-organic framework*, Nature 402 (1999) 276-279.

Ockwig, et al., Reticular chemistry: Occurrence and taxonomy nets and grammar for the design of frameworks, Acc. Chem. Res. 38(3) (2005) 176-182.

Seki, Dynamic channels of a porous coordination polymer responding to external stimuli, Phys. Chem. Chem. Phys. $4(10)$ (2002)

1968-1971.

Yaghi, et al., *Reticular synthesis and the design of new materials*,

Nature 423 (2003) 705-714.

PCT Feb. 24, 2011 International Search Report and Written Opinion

mailed in International Patent Application P

Chinese Patent Application 201080057358.8 (with English Translation).
CN Jul. 28, 2014 Response to First Office Action dated Mar. 14,

2014 filed in Chinese Patent Application 201080057358.8 (with English Translation).

EN Sep. 16, 2014 Notification of the Second Office Action mailed
in Chinese Patent Application 201080057358.8 (with English

Translation).

JP Nov. 26, 2013 Office Action mailed in Japanese Patent Application 2012-543346 (with English Translation).

JP Jul. 17, 2014 Examiner's Decision of Refusal mailed in Japanese

Patent Application 2012-54334

KR Feb. 10, 2014 Notice of Submission of Opinion/Office Action mailed in Korean Patent Application 10-2012-7013862 (with English Translation).
Den Exter, M.J., et al., Separation of permanent gases on the

all-silica 8-ring clathrasil DD3R, (Weitkamp, J., et al., eds. Zeolites and Related Microporous Materials: State of the Art 1994), Studies in Surface Science and Catalysis 84 (1994) 1159-66.

Gies, H., Studies on clathrasils. IX crystal structure of deca-
dodecasil 3R, the missing link between zeolites and clathrasils,

Zeitschrift für Kristallographie 175 (1986) 93-104.
Himeno, Shuji, et al., *Characterization and selectivity for methane* and carbon dioxide adsorption on the all-silica DD3R zeolite,

Micropor. Mesopor. Mater. 98 (2007) 62-69.
Tomita, Toshihiro, et al., *Gas separation characteristics of DDR* type zeolite membrane, Micropor. Mesopor. Mater. 68 (2004) 71-75. Yang, Qi-Liang, et al., Synthesis of DDR-type zeolite in fluoride medium, Chinese Journal of Inorganic Chemistry 25(2) (2009) 191-194.

Aguado, Sonia, et al., Facile synthesis of an ultramicroporous MOF tubular membrane with selectivity towards CO2, New J. Chem. 35 (2011) 41-44.

Bae, Tae-Hyun, et al., A high-performance gas-separation membrane containing submicrometer-sized metal-organic framework crystals, Angew. Chem. Int. Edit. 49 (2010) 9863-66.

Baker, Richard W., Future directions of membrane gas separation
technology, Ind. Eng. Chem. Res. 41(6) (2002) 1393-411.
Baker, Richard W. and Kaaeid Lokhandwala, *Natural gas process-*
ing with membranes: An overview, Ind.

Brar, Tejinder, et al., Control of crystal size and distribution of zeolite A, Ind. Eng. Chem. Res. 40 (2001) 1133-39.

Bux, Helge, et al., Zeolitic imidazolate framework membrane with molecular sieving properties by microwave-assisted solvothermal synthesis, J. Am. Chem. Soc. 131(44) (2009) 16000-01.

Caro, Jüergen and Manfred Noack, Zeolite membranes-Recent developments and progress, Micropor. Mesopor. Mater. 115 (2008) 215-33.

Carreon, Moises A., et al., Alumina-supported SAPO-34 membranes for CO_2/CH_4 separation, J. Am. Chem. Soc. 130(16) (2008) 5412-13.

Centrone, Andrea, et al., Growth of metal-organic frameworks on polymer surfaces, J. Am. Chem. Soc. 132(44) (2010) 15687-91.

Chen, Banglin, et al., A microporous metal-organic framework for gas-chromatographic separation of alkanes, Agnew. Chem. Int. Ed.

45 (2006) 1390-93.
Chiu, W. V., et al., *Post-synthesis defect abatement of inorganic membranes for gas separation*, J. Membr. Sci. 377 (2011) 182-90.
Cui, Ying, et al., *Preparation and gas separation performance of zeol*

Cui, Ying, et al., Preparation and gas separation properties of zeolite T membrane, Chem. Comm. 17 (2003) 2154-55.

OTHER PUBLICATIONS

Favre, Eric, Carbon dioxide recovery from post-combustion processes: Can gas permeation membranes compete with absorption?, J. Membr. Sci. 294 (2007) 50-59.
Forster, Paul M., et al., A high-throughput investigation of the role

of pH , temperature, concentration, and time on the synthesis of hybrid inorganic-organic materials, Angew. Chem. Int. Ed. 44 (2005) 7608-11.

Ge, Qingqin, et al., High-performance zeolite NaA membranes on polymer-zeolite composite hollow fiber supports, J. Am. Chem. Soc. 131(47) (2009) 17056-57.
Hao, Ji Hua and Shichang Wang, *Influence of quench medium on the*

structure and gas permeation properties of cellulose acetate mem-
branes, J. Appl. Polym. Sci. 68(8) (1998) 1269-76.

Heng, Samuel, et al., Low-temperature ozone treatment for organic template removal from zeolite membrane, J. Membr. Sci. 243 (2004) 69-78.

Himeno, Shuji, et al., Synthesis and permeation properties of a DDR-type zeolite membrane for separation of CO , $/CH₄$ gaseous

mixtures, Ind. Eng. Chem. Res. $46(21)$ (2007) 6989-97.
Huang, Aisheng, & Jürgen Caro, *Covalent post-functionalization of* zeolitic imidazolate framework ZIF-90 membrane for enhanced
hydrogen selectivity, Angew. Chem. Int. Ed. 50 (2011) 4979-82.

Huang, Aisheng, et al., Molecular-sieve membrane with hydrogen permselectivity: $ZIF-22$ in LTA topology prepared with

3-aminopropyltriethoxysilane as covalent linker, Angew. Chem. Int.
Ed. 49 (2010) 4958-61.
Huang, Aisheng and Jürgen Caro, *Cationic polymer used to capture*
zeolite precursor particles for the facile synthesis of oriented LTA molecular sieve membrane, Chem. Mater. $22(15)$ (2010) 4353-

Husain, Shabbir, Mixed Matrix Dual Layer Hollow Fiber Membranes for Natural Gas Separation, Dissertation, Georgia Institute of Technology, Atlanta, GA, 2006, 48-49.

Jee, Sang Eun and David S. Sholl, Carbon dioxide and methane transport in DDR zeolite: Insights from molecular simulations into carbon dioxide separations in small pore zeolites, J. Am. Chem. Soc. 131(22) (2009) 7896-7904.
Jie, Xingming, et al., *Gas permeation performance of cellulose*

 h ollow fiber membranes made from the cellulose/Nmethylmorpholine-N-oxide/ H_2O system, J. Appl. Polym. Sci. 91(3) (2004) 1873-80.

Kanezashi, Masakoto, et al., Gas permeation through DDR-type zeolite membranes at high temperatures, AIChE J. 54(6) (2008) 1478-86.

Koros, William J. and Rajiv Mahajan, Pushing the limits on possibilities for large scale gas separation: Which strategies?, J.

mossibilities for large scale gas separation in Membr. Sci. 175 (2000) 181-96.
Kuhn, Jelan, et al., *Detemplation of DDR type zeolites by ozonica-*
tion, Micropor. Mesopor. Mater. 120 (2009) 12-18.

Kumar, P. et al., Ordered mesoporous membranes: Effects of support and surfactant removal conditions on membrane quality, J. Membr.

Sci. 279 (2006) 539-47.
Kusakabe, Katsuki, et al., Formation of a Y-type zeolite membrane
on a porous alpha-alumina tube for gas separation, Ind. Eng.

Chem. Res. 36(3) (1997) 649-55.
Lee, Clare, et al., *Thermodynamic and kinetic factors in the hydro-thermal synthesis of hybrid frameworks: Zinc 4-cyclohexene-1,2-*
dicarboylates, Chem. Comm. (2006) 2687-89.

Li, Shiguang, et al., Scale-up of SAPO-34 membranes for CO_2/CH_4
separation, J. Membr. Sci. 352 (2010) 7-13.

Li, Yan-Shuo, et al., Molecular sieve membrane: Supported metal-
organic framework with high hydrogen selectivity, Angew. Chem.

Int. Edit. 49 (2010) 548-51.
Li, Zong-Qun, et al., Fabrication of nanosheets of a fluorescent
metal-organic framework [Zn(BDC)(H,O)]. (BDC=1.4framework $[Zn(BDC)(H_2O)]_n$ benzenedicarboxylate): Ultrasonic synthesis and sensing of ethylamine, Inorganic Chemistry Communications 11(11) (2008) 1375 - 77 .

Li, Zong-Qun, et al., Ultrasonic synthesis of the microporous metal-organic framework $Cu₃(BTC)$, at ambient temperature and pressure: An efficient and environmentally friendly method, Mater.
Lett. 63 (2009) 78-80.
Lin, Zhoujia, et al., Chiral induction in the ionothermal synthesis of

a 3-D coordination polymer, J. Am. Chem. Soc. 129(16) (2007) 4880-81.

Lindmark, Jonas and Jonas Hedlund, Modification of MFI mem-
branes with amine groups for enhanced CO_2 selectivity, J. Mater. Chem. 20(11) (2010) 2219-25.
Liu, Xin-Lei, et al., An organophilic pervaporation membrane

derived from metal-organic framework nanoparticles for efficient recovery of bio-alcohols, Angew. Chem. Int. Ed. $50(45)$ (2011) 10636-39.

Liu, Yunyang, et al., Synthesis and characterization of ZIF-69 membranes and separation for CO_2/CO mixture, J. Membr. 353 (2010) 36-40.

Liu, Yunyang, et al., Synthesis of highly c-oriented ZIF-69 mem-
branes by secondary growth and their gas permeation properties, J.

brand Membr. Sci. 379 (2011) 46-51.

Ma, Bao-Qing, et al., *Microporous pillared paddle-wheel frame-*
 works based on mixed-ligand coordination of zinc Ions, Inorg.

Chem. 44(14) (2005) 4912-14.
Matsuda, H., et al., *Improvement of ethanol selectivity of silicalite* membrane in pervaporation by silicone rubber coating, J. Membr.

Sci. 210(2) (2002) 433-37.
McCarthy, Michael C., et al., *Synthesis of zeolitic imidazolate* framework films and membranes with controlled microstructures,

Langmuir 26(18) (2010) 14636-41.
Morris, William, et al., Crystals as molecules: Postsynthesis cova-
lent functionalization of zeolitic imidazolate frameworks, J. Am.

chem. Soc. 130(38) (2008) 12626-27.
Nair, Sankar, et al., Separation of close boiling hydrocarbon mix-
tures by MFI and FAU membranes made by secondary growth, Micropor. Mesopor. Mater. 48 (2001) 219-28.
Pachfule, Pradip, et al., Solvothermal synthesis, structure, and

properties of metal organic framework isomers derived from a partially fluorinated link, Cryst. Growth Des. 11 (2011) 1215-22.

Pan, Long, et al., Microporous metal organic materials: Promising candidates as sorbents for hydrogen storage, J. Am. Chem. Soc. 126(5) (2004) 1308-09.

Parikh, Atul N., et al., Non-thermal calcination by ultraviolet irradiation in the synthesis of microporous materials, Micropor.
Mesopor. Mater. 76 (2004) 17-22.
Qiu, Wulin, et al., *Dehydration of ethanol-water mixtures using*

asymmetric hollow fiber membranes from commercial polyimides, J.
Membr. Sci. 327 (2009) 96-103.
Ranjan, Rajiv and Michael Tsapatsis, Microporous metal organic
framework membrane on porous support using the seeded growth

method, Chem. Mater. xxx(xx) (2009) 000-000.
Sommer, Stefan and Thomas Melin, Influence of operation parameters on the separation of mixtures by pervaporation and vapor permeation with inorganic membranes. Part 1: Dehydration of solvents, Chem. Eng. Sci. 60 (2005) 4509-23.

Son, Won-Jin, et al., *Sonochemical synthseis of MOF-5*, Chem.
Commun. 47 (2008) 6336-38.
Van Den Bergh, J., et al., *Separation and permeation characteristics* of a DD3R zeolite membrane, J. Membr. Sci. 316 (2008) 35-45.

Van Den Bergh, Johan, et al., *Modeling permeation of* CO_2/CH_4 , N_2/Ch_4 , and CO_2/air mixtures across a DD3R zeolite membrane, J. Phys. Chem. C 114(20) (2010) 9379-89.
Venna, Surendar R. and Moises A. Carreon, *Highly*

tion, J. Am. Chem. Soc.132(1) (2010) 76-78.
Watanabe, Taku, et al., Computational identification of a metal organic framework for high selectivity membrane-based CO , / $CH₄$ separations: $Cu(hfipbb)(H_2hfipbb)_{0.5}$, Phys. Chem. Chem. Phys. 11 (2009) 11389-94.

Xomeritakis, George, et al., Transport properties of alumina-supported MFI membranes made by secondary (seeded) growth, Micropor. Mesopor. Mater. 38 (2000) 61-73.

OTHER PUBLICATIONS

Weh, K., et al., *Permeation of single gases and gas mixtures through faujasite-type molecular sieve membrances*, Micropor. Mesopor. Mater. 54 (2002) 27-36.
Xu, Gengsheng, et al., *Preparation of ZIF-8 membranes supported*

Membr. Sci. 385-386 (2011) 187-93.
Yao, Jianfeng, et al., *Contra-diffusion synthesis of ZIF-8 films on a polymer substrate*, Chem. Commun. 47 (2011) 2559-61.

Yan, Yushan, et al., Preparation of highly selective zeolite ZSM-5 membranes by a post-synthetic coking treatment, J. Membr. Sci.

123(1) (1997) 95-103.

Yang, Lisha and Huimin Lu, Microwave-assisted ionothermal synthesis and characterization of zeolitic imidazolate framework-8,

Chinese J. Chem. 30 (2012) 1040-44.

Yoo, Won Cheol, et al., *High-perf*

Angew. Chem. Int. Ed. 49 (2010) 8699-703.
Zhang, Yanfeng, et al., *Blocking defects in SAPO-34 membranes* with cyclodextrin, J. Membr. Sci. 358 (2010) 7-12.

Zheng, Zhenkun, et al., Synthesis, characterization and modifica-
tion of DDR membranes grown on alpha-alumina supports, J.

tion of DDR membranes grown on alpha-alumina supports, J.
Mater. Sci. 43 (2008) 2499-502.
PCT Jul. 27, 2012 International Search Report and Written Opinion
issued in International Patent Application PCT/US2012/025064.
PCT

U.S. Appl. No. 13/611,988, filed Sep. 12, 2012.
Jun. 14, 2013 Response to Office Action/Noncompliant Amendment

dated Jun. 10, 2013 for U.S. Appl. No. $13/611,988$, filed Sep. 12, 2012.
Sep. 5, 2013 Office Action mailed for U.S. Appl. No. $13/611,988$,

filed Sep. 12, 2012.
Nov. 5, 2013 Response to Final Office Action dated Sep. 5, 2013 for
U.S. Appl. No. 13/611,988, filed Sep. 12, 2012.

Nov. 18, 2013 Notice of Allowability mailed for U.S. Appl. No. 13/611,988, filed Apr. 4, 2011.

Nov. 20, 2013 Notice of Allowability mailed for U.S. Appl. No.
13/399,645, filed Feb. 17, 2012.
Marler, et al., *Studies on clathrasils VIII. Nonasils*-[4¹5⁸],
88SiO₂*8M⁸*8M⁹*4M²⁰ : *Synthesis, thermal propert*

Aksay, et al., *Biometric pathways for assembling inorganic thin films*, Science 273(5277) (1996) 892-898.

Miyata, et al., Silica films with a single-crystalline mesoporous
structure, Nat. Mater. 3(9) (2004) 651-656.

Yang, et al., *Registered growth of mesoporous silica films on graphite*, J. Mater. Chem. 7(7) (1997) 1285-1290.

JP Jun. 17, 2014 Examiner's Decision of Refusal mailed in Japanese
Patent Application 2012-543346 (with English Translation).

CN Mar. 9, 2015 Response to Office Action (with English translation of claims) filed in China Patent Application No. 201180052181.

JP Apr. 7, 2015 Office Action mailed in Japan Patent Application No. 2013-536619 (with English translation).

JP Apr. 24, 2015 English translation of Description and Claims for JP Patent Application Laid-Open No. 2002-338229.

JP Apr. 24, 2015 English translation of Description and Claims for JP Patent Application Laid-Open No. 2008-173576.

CN Dec. 2, 2014 Response to Office Action filed in China Patent Application No. 201080057358.8 (with English translation).

CN Mar. 16, 2015 Third Office Action mailed in China Patent Application No. 201080057358.8 (with English translation).

Sep. 1, 2015 Office Action mailed in U.S. App. No. 13/897,939, filed May 20, 2013.
Dec. 1, 2015 Amendment and Response to Office Action dated Sep. 1, 2015 in U.S. Appl. No. 13/897,939, filed May 20, 2013.
JP Dec. 15, 2015

Brown, et al., Interfacial microfluidic processing of metal-organic
framework hollow fiber membranes, Science 345(6192) (2014)
72.75

Calvo, et al., *Pore size distributions in macroporous membranes*, *J.* Colloid & Interface Sci. $176(2)$ (1995) 467-78.
Khare, et al., *A quantitative method for measuring nanocomposite dispersion*, Polymer 51 (2010) 719

MFI membranes by permporometry, J. Membr. Sci. 417-418 (2012) 183-92, Accepted manuscript (2015).

Zhang, et al., Unexpected molecular sieving properties of Zeolitic Imidazolate Framework-8, J. Phys. Chem. Lett. 3(16) (2012) 2130-34.

Feb. 2, 2016 Final Office Action mailed in U.S. Appl. No. 13/897,939, filed May 20, 2013.
Mar. 15, 2016 Amendment and Response to Final Office Action

dated Feb. 2, 2016; and First Request for Continued Examination filed in U.S. Appl. No. 13/897,939, filed May 20, 2013.

Mar. 22, 2016 Notice of Allowance / Notice of Allowability mailed in U.S. Appl. No. 13/897,939, filed May 20, 2013.

ES.

FIG.6A

EIC. TB

njo, 7A

mo. 8A

Ajsuejuj

FC.16

EG. ITB

HG. ITA

FIG. ISC

FIG. 18B

FIG.I8D

FIG.20B

ERG. 20D

FIG. 20C

FIG.25B

FIG.25A

FIG. 28

FIG. 33

patent application Ser. No. 14/231,871, which claims prior-
ity to U.S. Provisional Patent Application Ser. No. 61/913, ¹⁰ growth of ZIF-90 membranes on the outer surfaces of ity to U.S. Provisional Patent Application Ser. No. 61/913, ¹⁰ growth of ZIF-90 membranes on the outer surfaces of 592, filed on Dec. 9, 2013 for "Flow Processing and Char-
porous polymeric poly(amide-imide) (e.g., TORL acterization of Metal-Organic Framework (MOF) hollow fibers (about 250 μ m outer diameter) by a seeded
Membranes in Hollow Fiber and Tubular Modules" and to arough process¹² at mild conditions (65° C in methanol Membranes in Hollow Fiber and Tubular Modules" and to growth process¹² at mild conditions (65° C. in methanol U.S. Provisional Patent Application Ser. No. 61/820,489, ₁₅ solutions). Molecular sieving membranes on the

FEDERALLY SPONSORED RESEARCH while avoiding membrane-membrane contact
STATEMENT 20 interfaces that lead to defects during synthesis.

 N/A

This invention relates to a scalable, in situ flow-process- 30 ing method to synthesize metal-organic framework (MOF) Thus, an Interfacial Microfluidic Membrane Processing
membranes on or inside hollow fiber and tubular modules (IMMP) approach for preparing defect-controlled and membranes on or inside hollow fiber and tubular modules (IMMP) approach for preparing defect-controlled and and to characterize their molecular transport properties, and, defect-free molecular sieving membranes on the inne and to characterize their molecular transport properties, and, defect-free molecular sieving membranes on the inner sur-
in particular, to a high-efficiency, high-performance zeolitic faces of microscale hollow fibers is n imidazolate framework (ZIF)-8 mixed material that is free of 35 performance in gas and liquid separations.
defects. SUMMARY OF THE INVENTION
BACKGROUND OF THE INVENTION

est as high-performance separation systems for production separate module. However, in this invention, the inventors of clean and renewable fuels, building block chemicals, and demonstrate an in situ flow process to synthe of clean and renewable fuels, building block chemicals, and demonstrate an in situ flow process to synthesize a mem-
specialty chemicals. Compared to thermodynamically-
brane on or inside hollow fiber and tubular modules a specialty chemicals. Compared to thermodynamically-
diven separation methods, membrane-based processes can
controlled location, and to subsequently activate the memsignificantly reduce the energy and capital costs of separat-45 ing molecules on a large scale. For example, energy-intening molecules on a large scale. For example, energy-inten-
sive methods such as cryogenic distillation are commonly membranes during module construction and handling are sive methods such as cryogenic distillation are commonly membranes during module construction and handling are
used to separate hydrocarbons because of their quite similar eliminated. Lastly, this reactor cell design is ca thermodynamic properties. Membranes composed of providing continuous flow in the bore and shell side of the molecular sieving materials such as zeolites,¹ layered zeo- 50 hollow fiber or tubular module to allow for reage molecular sieving materials such as zeolites,¹ layered zeo- 50 hollow fiber or tubular module to allow for reagent recycling lites,² or metal-organic frameworks³ (MOFs) have intrinsic and to homogenize membrane th advantages over polymeric membranes such as a simulta-

By using a single module for synthesis and characterization,

neously high permeability and selectivity. Despite their

performance limitations, polymeric membranes h tinued to dominate industrial membrane separations due to 55 acterization of the permeation properties of the module their relative ease of processing into morphologies such as each treatment step to determine cause and ef hollow fibers.⁴ The greatest scientific challenge facing in other words, this reactor cell design facilitates a scal-
molecular sieving membranes is the lack of an easily scal-
able and generalizable method of processing molecular sieving membranes is the lack of an easily scal-
able and generalizable method of processing molecular
able, reliable, and benign fabrication process.⁵ This limita-
sieving membranes (specifically, ZIF-8 membra tion has been particularly severe for zeolite membranes, ω referred to as an Interfacial Microfluidic Membrane Pro-
which are typically fabricated by hydrothermal synthesis on essing (IMMP) approach. The IMMP approach c which are typically fabricated by hydrothermal synthesis on cessing (IMMP) approach. The IMMP approach combines high-cost support materials.
three key concepts: first, a two-solvent interfacial approach

reducing the barriers to economical zeolitic membranes, $6-7$ brane formation (at inner and outer surfaces, as well as the advent of metal-organic framework (MOF) molecular 65 inside the bulk, of the porous fiber); seco sieves has created potential for more scalable membrane ishment, and recycling of reactants at microfluidic condi-
fabrication processes under relatively benign conditions.⁸ tions in the hollow fiber bore; and third, mem

 $\mathbf{2}$

HIGH EFFICIENCY, HIGH PERFORMANCE
METAL-ORGANIC FRAMEWORK (MOF) bonds to organic linker molecules, and have been used to **METAL-ORGANIC FRAMEWORK (MOF)** bonds to organic linker molecules, and have been used to **MEMBRANES IN HOLLOW FIBERS AND** grow crystalline membranes through techniques similar to NES IN HOLLOW FIBERS AND grow crystalline membranes through techniques similar to
TUBULAR MODULES those developed for zeolitic membranes. The zeolitic imithose developed for zeolitic membranes. The zeolitic imi-
5 dazolite framework (ZIF) subclass of MOFs is of particular FRIOR RELATED APPLICATIONS interest for membrane fabrication, because of its tunable pore size and chemistry, $\frac{9}{2}$ and relatively good thermal and The present application is a continuation-in-part of U.S. chemical stability.¹⁰⁻¹¹ In an early demonstration of scalable patent application Ser. No. 14/231,871, which claims prior-
ZIF membrane processing, we recently d filed on May 7, 2013 for "Flow Processing and Character-

ization of Metal-Organic Framework (MOF) Membranes in

Tubular and Hollow Fiber Modules."

Tubular and Hollow Fiber Modules."

FEDER ALLY SPONSORED RESEARCU

It has been shown that free-standing MOF films and Not Applicable ("N/A") spheroids can be synthesized at the interfaces between two immiscible solvents.¹³ However, the growth of defect-con-REFERENCE TO MICROFICHE APPENDIX trolled membranes on the inner surfaces of microscale hollow fibers $(50-300 \,\mu m)$ inner (bore) diameter) is a key, and more challenging, advance. As the bore size (and hence volume) is decreased to microscopic dimensions, the like-
lihood of reactant depletion and local inhomogeneity FIELD OF INVENTION lihood of reactant depletion and local inhomogeneity
increase, leading to loss of control over membrane continu-
relates to a scalable, in situ flow-process- 30 ity and defect density.¹⁴

Zeolite and MOF membranes are typically synthesized in Molecular sieving membranes have generated great inter-40 a reactor, and then removed for washing and mounting in a controlled location, and to subsequently activate the membrane and measure separation properties in situ without eliminated. Lastly, this reactor cell design is capable of controlled accurately. This device also allows in situ characterization of the permeation properties of the module after

sieving membranes (specifically, ZIF-8 membranes) referred to as an Interfacial Microfluidic Membrane Pro-While substantive progress is being made in gradually that can be tuned to achieve positional control over mem-
ducing the barriers to economical zeolitic membranes, $6-7$ brane formation (at inner and outer surfaces, as tions in the hollow fiber bore; and third, membrane synthesis

in situ directly in the membrane module, which also func-
tions as a membrane synthesis reactor.
about 10 mol/L 2-methyl imidazole (mIm) in deionized

isothermal and initial heating IMMP approach that leads to In an embodiment, the shell solution contains an excess a mechanism of ZIF-8 membrane growth in hollow fibers ⁵ mIm ligand concentration ranging from about 1.2 t ing of ZIF-8 hollow fiber membrane formation can be used comprises the step of gently stirring the shell solution to
to engineer a higher-throughput, higher-selectivity ZIF-8¹⁰ prevent formation of local concentration gr membrane that is free of defects for propane/propylene embodiment, the shell solution is stirred at about 40 rpm to separations. Further, the inventors present detailed binary about 80 rpm. separations. Further, the inventors present detailed binary about 80 rpm.
and ternary mixed gas permeation measurements to high-
light the excellent separation characteristics of these defect-
 $\frac{1}{2}$ further comprises t light the excellent separation characteristics of these defect-
controlled and defect-free ZIF-8 hollow fiber membranes.

molecular sieving membranes comprises a hollow fiber having a first end and a second end, wherein a length of the first end is supported by and sealed into the first hole and a $_{20}$ flow rate for a third period; and stopping the second flow rate length of the second end is supported by and sealed into the of the bore solution for a second hole; wherein the first and second ends of the hollow
the second temperature is decreased to a third temperature
fiber are capped with a capping solution, wherein a molecu-
lar sieving membrane that is uniform and a of defects is grown on an inner bore surface of the hollow 25 about 1 μ L/hour to about 1000 μ L/hour per individual fiber. In an embodiment, the molecular sieving membrane is hollow fiber. In an embodiment, the first fiber. In an embodiment, the molecular sieving membrane is hollow fiber. In an embodiment, the first and second flow at least about 97% free of defects and wherein any defects rates are about 10 μ L/hour to about 100 $\$ at least about 97% free of defects and wherein any defects rates are about 10 μ L/hour to about 100 μ L/hour per indi-
are less than about 20 nm in diameter, length or width. In an vidual hollow fiber. are less than about 20 nm in diameter, length or width. In an vidual hollow fiber.

embodiment, the molecular sieving membrane is at least In an embodiment, the first period is about 1 hour to about 4 hours, about 97% free

In an embodiment, the lengths of the first and second ends
of the fourth period is about 3 hours to about 4 hours.
of the hollow fiber are sealed into the first and second holes
with an adhesive.
further comprises the step

In an embodiment, the capping solution is about 8 wt % 35 solvents selected from the group consisting of 1-octanol,
to about 10 wt % poly(dimethylsiloxane) (PDMS) in hep-
tane. In an embodiment, a method of making a reacto

In an embodiment, a method of making a reactor cell comprises the steps of: fluidically connecting a feed mixture
comprises the steps of: fabricating a reactor chamber extend-
to the first inlet: fluidically connecting a s ing into the reactor module from the upper surface of the 40 second inlet; collecting a separated mixture from the reactor module; fabricating an O-ring groove with an inner outlet; and collecting permeate from the second dimension slightly larger than and offset from the outer In an embodiment, the feed mixture to the reactor cell is dimension of the reactor chamber; fabricating the first hole selected from the group consisting of hydrogen extending into the reactor chamber from the first surface and
the second hole opposing the first hole and extending into 45 butanes. In an embodiment, the feed mixture to the reactor the second hole opposing the first hole and extending into 45 butanes. In an embodiment, the feed mixture to the reactor the reactor chamber from the second surface; fabricating a cell is selected from the group consisting the reactor chamber from the second surface; fabricating a cell is selected from the group consisting of hydrogen/
third hole extending into the reactor chamber from a third propane, propylene/propane and butenes/butanes. third hole extending into the reactor chamber from a third propane, propylene/propane and butenes/butanes. In an surface and a fourth hole opposing the third hole and embodiment, the feed mixture to the reactor cell compri extending into the reactor chamber from a fourth surface; about 2 mol % to about 95 mol % i-butane, about 2 mol % supporting and sealing the length of the first end of the 50 to about 95 mol % n-butane and mixtures thereof hollow fiber in the first hole and the length of the second end
of the ambodiment, an operating temperature for the
of the hollow fiber into the second hole; capping the first and
reactor cell is from about 30° C, to abou of the hollow fiber into the second hole; capping the first and reactor cell is from about 30° C. to about 95° C. or any value second ends of the hollow fiber with a capping solution; there between. fabricating the reactor module cover to fit on the upper In an embodiment, an operating pressure for the reactor is surface of the reactor module and fastening the reactor 55 from about 1 bar to about 14 bar or any value t module cover to the seal the reactor chamber; fluidly con-
nese and other objects, features and advantages will
necting a bore solution to the first inlet; fluidly connecting a
become apparent as reference is made to the f

 Zn^{+2} concentration ranging from about 0.005 to about 0.1 mol/L Zn^{+2} in 1-octanol. In an embodiment, the bore solution contains a limited Zn^{+2} concentration ranging from The patent or application file contains at least one drawing about 0.01 to about 0.03 mol/L Zn^{+2} in 1-octanol. In an 65 executed in color. Copies of

the present invention, the inventors demonstrate an under the present invention, the inventors demonstrate an under.

a mechanism of ZIF-8 membrane growth in hollow fibers $\frac{5}{1.6}$ mIm ligand concentration ranging from about 1.2 to about that elucidates the roles of reactant transport. film crystal-
1.6 mol/L 2-methyl imidazole (mIm) 1.6 mol/L 2-methyl imidazole (mIm) in deionized water. In lization, and the hollow fiber support microstructure. The an embodiment, the shell solution is about 1.37 mol/L mIm inventors also demonstrate how a mechanistic understand-
in deionized water. In an embodiment, the method inventors also demonstrate how a mechanistic understand-
in deionized water. In an embodiment, the method further
ing of ZIF-8 hollow fiber membrane formation can be used
comprises the step of gently stirring the shell sol to engineer a higher-throughput, higher-selectivity $ZIF-8$ ¹⁰ prevent formation of local concentration gradients. In an membrane that is free of defects for propane/propylene embodiment, the shell solution is stirred at

ntrolled and defect-free ZIF-8 hollow fiber membranes. First flow rate for a first period, wherein a first temperature
In an embodiment, a reactor cell for flow processing is increased to a second temperature during a port is increased to a second temperature during a portion of the first period; stopping the first flow rate of the bore solution for a second period; flowing the bore solution at a second

than about 10 nm in diameter, length or width.
In an embodiment, the lengths of the first and second ends
the fourth period is about 3 hours to about 4 hours.

th an adhesive.
In an embodiment, the capping solution is about 8 wt $\%$ 35 solvents selected from the group consisting of 1-octanol.

to the first inlet; fluidically connecting a sweep gas to the second inlet; collecting a separated mixture from the first

shell solution to the second inlet; and growing a molecular detailed description, preferred embodiments, and examples, sieving membrane that is uniform and at least about 95% given for the purpose of disclosure, and taken

embodiment, the bore solution is about 0.018 mol/L Zn^{+2} in publication with color drawing(s) will be provided by the 1-octanol. In an embodiment, the shell solution contains an Office upon request and payment of the n Office upon request and payment of the necessary fee.

following detailed disclosure, taken in conjunction with the conditions; accompanying drawings, in which like parts are given like FIG. 13 illustrates a chart of Time (min.) vs. Flow Rate reference numerals, and wherein:

TORLON®) hollow fiber epoxy-sealed inside a reactor mounted in a reactor cell, (B) depicts an exploded side view
according to an embodiment of the present invention: of hollow fiber support during synthesis, showing $\text{Z$

side surface of a seeded polyamide-imide (e.g., TORLON®) hollow fiber:

FIG. 5B illustrates a top-view of a SEM micrograph of a 25 ZIF-8 membrane grown on outer (shell) side surface of a FIG. 16 illustrates a schematic of steady-state gas perme-

FIG. 6A illustrates a cross-sectional view of a SEM ing as the permeation cell;
icrograph of a ZIF-8 membrane grown on a polyamide-
FIG. 17A illustrates a cross-sectional view of a SEM micrograph of a ZIF-8 membrane grown on a polyamideimide (e.g., TORLON®) hollow fiber under aqueous flow 30 micrograph of a ZIF-8 membrane grown in polyamide-imide

FIG. 6B illustrates a detailed cross-sectional view of a surface by manipulating the location of the two reactants SEM micrograph of a ZIF-8 membrane grown on a poly- (i.e., zinc nitrate hexahydrate and mIm) and the two so amide-imide (e.g., TORLON®) hollow fiber under aqueous flow conditions;

FIG. 7A illustrates a cross-sectional view of a SEM (red) and superimposed zinc (green) showing the localizamicrograph of a ZIF-8 membrane grown on a polyamide-
micrograph of a ZIF-8 membrane grown on a polyamide-
ion of t micrograph of a ZIF-8 membrane grown on a polyamide-
inner of the ZIF-8 membrane on the inner (bore) side surface
inide (e.g., TORLON®) hollow fiber as in imide (e.g., TORLON®) hollow fiber under 1-octanol flow of the polyamide-imide (e.g., TORLON®) hollow fiber as in conditions;

FIG. 7B illustrates a detailed cross-sectional view of a 40 SEM micrograph of ZIF-8 membrane grown on a poly-SEM micrograph of ZIF-8 membrane grown on a poly-
amide-imide (e.g., TORLON®) hollow fiber under 1-octa-
flow conditions used to synthesize ZIF-8 membranes on the amide-imide (e.g., TORLON®) hollow fiber under 1-octa-
flow conditions used to synthesize ZIF-8 membranes on the
not flow conditions;
 $\frac{1}{2}$ inner (bore) side surface of a polyamide-imide (e.g., TOR-

FIG. 8A illustrates a cross-sectional view of a SEM LON®) hollow fibers as in Example 1;
icrograph of a ZIF-8 membrane grown on a polyamide- 45 FIG. 18B illustrates a cross-sectional view of a SEM micrograph of a ZIF-8 membrane grown on a polyamide-45 imide (e.g., TORLON®) hollow fiber using an Interfacial imide (e.g., TORLON®) hollow fiber using an Interfacial micrograph of a ZIF-8 membrane grown in a polyamide-
Microfluidic Membrane Processing (IMMP) approach with imide (e.g., TORLON®) hollow fiber using the IMMP Microfluidic Membrane Processing (IMMP) approach with imide (e.g., TORLON®) hollow fiber using the IMMP 2 mL/min flow rate of a $\text{Zn}^{+2}/1$ -octanol bore solution and an approach with a static flow of a $\text{Zn}^{+2}/1$ -oct 2 mL/min flow rate of a $\text{Zn}^{+2}/1$ -octanol bore solution and an approach with a static flow of a $\text{Zn}^{+2}/1$ -octanol bore solu-
aqueous 2-methyl imidazole (mIm) shell solution according tion, depicting a discontinuo

FIG. 8B illustrates a cross-sectional view of a SEM micrograph of a ZIF-8 membrane grown on a polyamidemicrograph of a ZIF-8 membrane grown on a polyamide-
imide (e.g., TORLON®) hollow fiber using the IMMP imide (e.g., TORLON®) hollow fiber using the IMMP imide (e.g., TORLON®) hollow fiber using the IMMP imide (e.g., TORLON®) hollow fiber using the IMMP approach with 1 $\mu L/min$ flow rate of a $\text{Zn}^{+2}/1$ -octanol bore approach with a continuous flow of a $\text{Zn}^{+2}/1$ -octan solution and an aqueous mIm shell solution according to an 55 embodiment of the present invention;

ZIF-8 membranes, confirming structure of supported ZIF-8 membranes;

FIG. 10 illustrates an Energy Dispersive X-ray Spectros-
copy (EDS) Zinc line profile scan chart of Distance from copy (EDS) Zinc line profile scan chart of Distance from um thick) on the inner (bore) side surface of the hollow fiber;
membrane surface (um (microns)) vs. Intensity, depicting FIG. 19 illustrates an XRD pattern confirmin

and continuous flow conditions, showing static growth con-
titlent flow of a $Zn^{+2}/1$ -octanol bore solution as in
ditions;
Example 1;

For a further understanding of the nature and objects of FIG. 12 illustrates a chart of Time vs. Flow Rate for the present invention, reference should be made to the pulsed flow conditions, showing growth and replenishment

FIG. 1A illustrates a side perspective view of a hollow

FIG. 1A illustrates a side perspective view of a hollow

for or tubular membrane module showing location of a

bore and a shell;

FIG. 1B illustrates a cross-section

a reactor according to an embodiment of the present inven-
tion;
FIG. 15 illustrates a schematic of the IMMP approach: (A)
FIG. 2 illustrates a photograph of a polyamide imide (e.g. 15 depicts a side view of a plurality o FIG. 3 illustrates a photograph of a polyamide-imide (e.g., 15 depicts a side view of a plurality of hollow fibers (orange)
DRI ON®) hollow fiber enoxy-sealed inside a reactor mounted in a reactor cell, (B) depicts an e according to an embodiment of the present invention;
FIG 4 illustrates a photograph of a reactor connected to being supplied by a 1-octanol solution (light red) flowing FIG. 4 illustrates a photograph of a reactor connected to
cluster of the beng supplied by a 1-octanol solution (light red) flowing
cycling peristaltic pump according to an embodiment of the
 $\frac{1}{20}$ linkers being suppli $p₂₀$ linkers being supplied on the outer (shell) side of the hollow FIG. 5A illustrates a cross-sectional view of a SEM fiber in an aqueous solution (light blue), and (C) depicts a micrograph of a ZIF-8 membrane grown on an outer (shell) further exploded view of hollow fiber support during further exploded view of hollow fiber support during synthesis, showing the membrane forming on the inner surface of the hollow fiber by reaction of the two precursors to form a polycrystalline ZIF-8 layer (dark blue);

seeded polyamide-imide (e.g., TORLON®) hollow fiber; ation measurement apparatus, with the reactor cell function-
FIG. 6A illustrates a cross-sectional view of a SEM ing as the permeation cell;

conditions;
FIG. 6B illustrates a detailed cross-sectional view of a surface by manipulating the location of the two reactants (i.e., zinc nitrate hexahydrate and mIm) and the two solvents (i.e., deionized water (DI) and 1-octanol) as in Example 1;

our conditions;
FIG. 17B illustrates an EDS elemental maps of carbon
FIG. 7A illustrates a cross-sectional view of a SEM (red) and superimposed zinc (green) showing the localiza-

Example 1;
FIG. 18A illustrates a chart of Time (minutes) vs. Flow inner (bore) side surface of a polyamide-imide (e.g., TOR-LON®) hollow fibers as in Example 1;

tion, depicting a discontinuous particle coating on the inner to an embodiment of the present invention;

FIG. 8B illustrates a cross-sectional view of a SEM

FIG. 18C illustrates a cross-sectional view of a SEM

FIG. 18C illustrates a cross-sectional view of a SEM

approach with a continuous flow of a $Zn^{+2}/1$ -octanol bore solution, depicting a continuous $ZIF-8$ membrane (about 3 abodiment of the present invention;

FIG. 9 illustrates a X-ray Diffraction (XRD) chart of FIG. 18D illustrates a cross-sectional view of a SEM

FIG. 9 illustrates a X-ray Diffraction (XRD) chart of FIG. 18D illustrates a cross-sectional view of a SEM 2Theta ($^{\circ}$) vs. Intensity for simulated and experimental micrograph of a ZIF-8 membrane grown in a polyamidemicrograph of a ZIF-8 membrane grown in a polyamide-
imide (e.g., TORLON®) hollow fiber using the IMMP 60 approach with an intermittent flow of a $Zn^{+2}/1$ -octanol bore solution, depicting a continuous $ZIF-8$ membrane (about 8)

thickness of membrane to be \sim 3 μ m; structure of ZIF-8 membrane grown on a polyamide-imide FIG. 11 illustrates a chart of Time vs. Flow Rate for static 65 (e.g., TORLON®) hollow fiber using the IMMP approach Example 1;

imide (e.g., TORLON®) hollow fiber closer to an inner etration of PDMS;
(bore) side surface by manipulating the location of the two FIG. 25B illustrates an EDS elemental map of Si (red) (bore) side surface by manipulating the location of the two FIG. 25B illustrates an EDS elemental map of Si (red) reactants (i.e., zinc nitrate hexahydrate and mIm) and the s showing the penetration of PDMS into the pores reactants (i.e., zinc nitrate hexahydrate and mIm) and the 5 two solvents (i.e., DI and 1-octanol) as in Example 2;

the solvents (i.e., DI and 1-octanol) as in Example 2; polyamide-imide (e.g., TORLON®) hollow fiber;

FIG. 2018 illustrates a cross-sectional view of a SEM FIG. 26A illustrates a schematic of the Interfaci FIG. 20B illustrates a cross-sectional view of a SEM FIG. 26A illustrates a schematic of the Interfacial Micromicrograph of a ZIF-8 membrane grown in a polyamide- fluidic Membrane Processing (IMMP) approach, showing imide (e.g., TORLON®) hollow fiber closer to an inner two feed streams for a ZIF-8 membrane on a polyamide-
(bore) side surface by manipulating the location of the two 10 imide (e.g., TORLON®) hollow fiber;
reactants (i.e reactants (i.e., zinc nitrate hexahydrate and mIm) and the two solvents (i.e., DI and 1-octanol) as in Example 3;

imide (e.g., TORLON®) hollow fiber closer to an outer 15 an inner surface of polyamide imide (e.g., TORLON®) (shell) side surface by manipulating the location of the two hollow fibers with the numbered temporal locations reactants (i.e., zinc nitrate hexahydrate and mIm) and the and #3) indicating the times where the ZIF-8 membrane two solvents (i.e., DI and 1-octanol) as in Example 4; growth were arrested;

two solvents (i.e., DI and 1-octanol) as in Example 4;
FIG. 20D illustrates a cross-sectional view of a SEM FIG. 20D illustrates a cross-sectional view of a SEM FIG. 27A illustrates an exemplary cross-sectional SEM micrograph of a ZIF-8 membrane grown in a polyamide- 20 image of a ZIF-8 membrane grown on a polyamide-imide micrograph of a ZIF-8 membrane grown in a polyamide- 20 image of a ZIF-8 membrane grown on a polyamide-imide
imide (e.g., TORLON®) hollow fiber closer to an outer (e.g., TORLON®) hollow fiber after 2 hours of continuous imide (e.g., TORLON®) hollow fiber closer to an outer (e.g., TORLON®) hollow fiber after 2 hours of continuous (shell) side surface by manipulating the location of the two membrane growth (Step 1) using isothermal IMMP reactants (i.e., zinc nitrate hexahydrate and mIm) and the two solvents (i.e., DI and 1-octanol) as in Example 5;

Permeance (G.P.U.) and Separation Factor, showing binary (e.g., TORLON®) hollow fiber after 2 hours of continuous H_2/C_3H_8 permeation characteristics for an equimolar feed membrane growth (Step 1) and 3.5 hours of stat mixture on a ZIF-8 membrane grown in a polyamide-imide growth (Step 2) using isothermal IMMP approach;
(e.g., TORLON®) hollow fiber using the IMMP system and FIG. 27C illustrates an exemplary cross-sectional SEM (e.g., TORLON®) hollow fiber using the IMMP system and FIG. 27C illustrates an exemplary cross-sectional SEM capping ends of the hollow fiber with poly(dimethylsilo- 30 image of a ZIF-8 membrane grown on a polyamide-imide capping ends of the hollow fiber with poly (dimethylsilo- 30

 C_3H_6/C_3H_8 permeation characteristics for an equimolar feed membrane growth followed by 3.5 hours of second static mixture on a ZIF-8 membrane grown in a polyamide-imide 35 membrane growth (Step 3) using isothermal IM mixture on a ZIF-8 membrane grown in a polyamide-imide 35 (e.g., TORLON®) hollow fiber using the IMMP system and approach; capping ends of the hollow fiber with PDMS; FIG. 27

permeation characteristics for an equimolar H_2/C_3H_8 feed 40 membrane mixture on a ZIF-8 membrane grown in polyamide-imide approach; mixture on a ZIF-8 membrane grown in polyamide-imide approach;
(e.g., TORLON®) hollow fiber using the IMMP system; FIG. 27E illustrates an exemplary cross-sectional SEM (e.g., TORLON®) hollow fiber using the IMMP system; FIG. 22B illustrates a chart of Temperature (° C.) vs.

permeation characteristics for an equimolar C_3H_8/C_3H_8 feed 45 mixture on a ZIF-8 membrane grown in polyamide-imide mixture on a ZIF-8 membrane grown in polyamide-imide growth (Step 2) using initial heating IMMP approach;
(e.g., TORLON®) hollow fiber using the IMMP system; FIG. 27F illustrates an exemplary cross-sectional is

Permeance (G.P.U.) and Separation Factor, showing binary (e.g., TORLON®) hollow fiber after 2 hours of continuous permeation characteristics for an equimolar H_2/C_3H_8 feed 50 membrane growth (Step 1), 3.5 hours of stat permeation characteristics for an equimolar H_2/C_3H_8 feed 50 membrane growth (Step 1), 3.5 hours of static membrane mixture on ZIF-8 membranes grown simultaneously in three growth (Step 2) and 20 minutes of second cont mixture on ZIF-8 membranes grown simultaneously in three growth (Step 2) and 20 minutes of second continuous (3) polyamide-imide (e.g., TORLON®) hollow fiber using membrane growth followed by 3.5 hours of second static (3) polyamide-imide (e.g., TORLON®) hollow fiber using the IMMP system and sealing the ends of the fibers with PDMS; approach;

Permeance (G.P.U.) and Separation Factor, showing binary defective ZIF-8 membrane layers grown on a polyamide-
permeation characteristics for an equimolar C_3H_6/C_3H_8 feed imide (e.g., TORLON®) hollow fiber after 120 m (3) polyamide-imide (e.g., TORLON®) hollow fiber using approach;
the IMMP approach and sealing the ends of the fibers with ω FIG. 29 illustrates a schematic of the ZIF-8 membrane

FIG. 24B illustrates a schematic showing suppression of 65 fiber (tan) during synthesis after 2 hours of continuous the bypass effect in FIG. 24A by capping the hollow fiber membrane growth (Step 1), (C) depicts a side per

FIG. 20A illustrates a cross-sectional view of a SEM FIG. 25A illustrates a SEM cross-section image of the micrograph of a ZIF-8 membrane grown in a polyamide-
inner-surface ZIF-8 membrane after end-capping by peninner-surface ZIF-8 membrane after end-capping by pen-

fluidic Membrane Processing (IMMP) approach, showing

o solvents (i.e., DI and 1-octanol) as in Example 3; and Temperature (\degree C.), showing bore flow rate and two FIG. **20**C illustrates a cross-sectional view of a SEM different temperature profiles (isothermal and initial FIG. 20C illustrates a cross-sectional view of a SEM different temperature profiles (isothermal and initial heating) micrograph of a ZIF-8 membrane grown in a polyamide-
for the IMMP approach to synthesize ZIF-8 membranes for the IMMP approach to synthesize ZIF-8 membranes on an inner surface of polyamide imide (e.g., TORLON®)

membrane growth (Step 1) using isothermal IMMP approach;

the solvents (i.e., DI and 1-octanol) as in Example 5;
FIG. 27B illustrates an exemplary cross-sectional SEM
FIG. 21A illustrates a chart of Temperature (° C.) vs. 25 image of a ZIF-8 membrane grown on a polyamide-imide image of a ZIF-8 membrane grown on a polyamide-imide m embrane growth (Step 1) and 3.5 hours of static membrane growth (Step 2) using isothermal IMMP approach;

zane) (PDMS);
FIG. 21B illustrates a chart of Temperature (° C.) vs. membrane growth (Step 1), 3.5 hr of static membrane FIG. 21B illustrates a chart of Temperature (° C.) vs. membrane growth (Step 1), 3.5 hr of static membrane Permeance (G.P.U.) and Separation Factor, showing binary growth (Step 2) and 20 minutes of second continuous growth (Step 2) and 20 minutes of second continuous membrane growth followed by 3.5 hours of second static

pping ends of the hollow fiber with PDMS;
FIG. 27D illustrates an exemplary cross-sectional SEM
FIG. 22A illustrates a chart of Temperature (° C.) vs. image of a ZIF-8 membrane grown on a polyamide-imide FIG. 22A illustrates a chart of Temperature (° C.) vs. image of a ZIF-8 membrane grown on a polyamide-imide
Permeance (G.P.U.) and Separation Factor, showing binary (e.g., TORLON®) hollow fiber after 2 hours of continuous (e.g., TORLON®) hollow fiber after 2 hours of continuous membrane growth (Step 1 using initial heating IMMP)

FIG. 22B illustrates a chart of Temperature (° C.) vs. image of a ZIF-8 membrane grown on a polyamide-imide
Permeance (G.P.U.) and Separation Factor, showing binary (e.g., TORLON®) hollow fiber after 2 hours of continuous (e.g., TORLON®) hollow fiber after 2 hours of continuous membrane growth (Step 1) and 3.5 hours of static membrane

g., TORLON®) hollow fiber using the IMMP system; FIG. 27F illustrates an exemplary cross-sectional SEM
FIG. 23A illustrates a chart of Temperature (° C.) v. image of a ZIF-8 membrane grown on a polyamide-imide image of a ZIF-8 membrane grown on a polyamide-imide (e.g., TORLON®) hollow fiber after 2 hours of continuous the IMMP system and sealing the ends of the fibers with membrane growth (Step 3) using initial heating IMMP PDMS;

FIG. 23B illustrates a chart of Temperature $(° \text{ C.})$ v. 55 FIG. 28 illustrates a magnified SEM image of dense and continuous membrane growth using isothermal IMMP

the IMMP approach and sealing the ends of the fibers with ω FIG. 29 illustrates a schematic of the ZIF-8 membrane
PDMS;
PDMS: FIG. 24A illustrates a schematic showing feed gas mol-
ules by depicts a side perspective view of a
ules by passing the ZIF-8 membrane through the hollow hollow fiber (tan) at 0 hours and (B) depicts a side perspececules bypassing the ZIF-8 membrane through the hollow hollow fiber (tan) at 0 hours and (B) depicts a side perspec-
fiber ends;
FIG. 24B illustrates a schematic showing suppression of 65 fiber (tan) during synthesis after the bypass effect in FIG. 24A by capping the hollow fiber membrane growth (Step 1), (C) depicts a side perspective ends with a PDMS film;
view of the ZIF-8 membrane (gray) grown on the hollow view of the ZIF-8 membrane (gray) grown on the hollow

membrane growth ($Step 1$) and 3.5 hours of static membrane growth (Step 2), (D) depicts a side perspective view of the ZIF-8 membrane (gray) grown on the hollow fiber (tan) The following detailed description of various embodiduring synthesis after 2 hours of continuous membrane $\frac{5}{10}$ ments of the present invention references the acco growth (Step 1), 3.5 hours of static membrane growth (Step drawings, which illustrate specific embodiments in which 2) and 20 minutes of second continuous membrane growth the invention can be practiced. While the illustrat followed by a final 3.5 hours of second static membrane embodiments of the invention have been described with growth (Step 3); and a (E) depicts a photograph of the two particularity, it will be understood that various oth reactants (i.e., zinc nitrite hexahydride and mIm) in the two

FIG. 30D illustrates an associated binary images for $_{20}$ estimation of the prosity of the image in FIG. 30C;

images of a ZIF-8 membrane grown on a polyamide-imide
(e.g., TORLON®) hollow fiber using isothermal IMMP process to synthesize a membrane on or inside a hollow fiber (e.g., TORLON®) hollow fiber using isothermal IMMP annoach as in Case $1 \cdot 25$

images of a ZIF-8 membrane grown on a polyamide-imide

(e.g., TORLON®) hollow fiber using initial heating IMMP

Lallaw Fiber and the permeation properties of the module Details

(µm) for a ZIF-8 membrane grown on a modified poly-
amide-imide (e.g., TORLON®) hollow fiber with the red
dotted lines indicating the average thickness of about 9.1 µm
comprises a shell **105** and a bore **110**. In an embodi dotted lines indicating the average thickness of about 9.1 μm comprises a shell 105 and a bore 110. In an embodiment, the for the ZIF-8 membrane;
for the ZIF-8 membrane;

FIG. 32C illustrates a chart of Length (mm) v.s Thickness 45 (um) for a ZIF-8 membrane grown on a modified poly-(μ m) for a ZIF-8 membrane grown on a modified poly-
amide-imide (e.g., TORLON®) hollow fiber using initial (bore) diameter of about 50 μ m to about 300 μ m. heating IMMP approach with the red dotted lines indicating Design Considerations/Machining Detail for Reactor Cell
the average thickness of about 5 µm for the ZIF-8 mem-
A reactor cell of the present invention may be cubic the brane; the about 50 rectangular-, cylindrical- or cylindrical-like shaped (e.g., \sim 50 rectangular-, cylindrical- or cylindrical-like shaped (e.g.,

membrane thickness (mol/m·s), showing helium permeabili-
ties (expressed as fluxxmembrane thickness) measured at height, and a reactor module cover 420 was fabricated to room temperature under dry and wetted conditions for ZIF-8 have the same base shape as the reactor module 200 and a
membranes grown on polyamide-imide (e.g., TORLON®) 55 second height. In an embodiment, the base shape may membranes grown on polyamide-imide (e.g., TORLON®) 55 second height. In an embodiment, the base shape may be hollow fibers as in Cases 1-3;
selected from the group consisting of square, rectangular,

FIG. 34A illustrates a chart of Temperature (C) vs. Per-
meance (G.P.U.) and Separation Factor, showing binary
 H_2/C_3H_8 and C_3H_6/C_3H_8 permeation characteristics for an include any metal, or any plastic compatible H_2/C_3H_8 and C_3H_6/C_3H_8 permeation characteristics for an include any metal, or any plastic compatible with an adhe-equimolar feed mixture on a ZIF-8 membrane grown on 60 sive (e.g., epoxy), alcohol solvents and wa equimolar feed mixture on a ZIF-8 membrane grown on ω sive (e.g., epoxy), alcohol solvents and water. In an embodi-
polyamide-imide (e.g., TORLON®) hollow fiber as in Case ment, the metal may be selected from the group polyamide-imide (e.g., TORLON®) hollow fiber as in Case 3;

meance (G.P.U.) and Separation Factor, showing ternary national, Inc.), and the like. In an embodiment, the plastic $H_2/C_3H_6/C_3H_8$ permeation characteristics for equimolar 65 may be selected from the group consisting o $H_2/C_3H_6/C_3H_8$ permeation characteristics for equimolar 65 may be selected from the group consisting of polyether feed mixtures on a ZIF-8 membrane grown on polyamide-
imide (e.g., DELRIN® (DuPont imide (e.g., DELRIN®

fiber (tan) during synthesis after 2 hours of continuous DETAILED DESCRIPTION OF EMBODIMENTS
membrane growth (Step 1) and 3.5 hours of static membrane OF THE INVENTION

the invention can be practiced. While the illustrative particularity, it will be understood that various other modi-
fications will be apparent to and can be readily made by solvents (i.e., DI and 1-octanol) after 15 minutes of inter-
facial contact in a glass vial;
scope of the invention. Accordingly, it is not intended that facial contact in a glass vial;
FIG. 30A illustrates a cross-sectional SEM image of a the scope of the claims appended hereto be limited to the FIG. 30A illustrates a cross-sectional SEM image of a the scope of the claims appended hereto be limited to the polyamide imide (e.g., TORLON®) hollow fiber; polyamide imide (e.g., TORLON®) hollow fiber;
FIG. 30B illustrates an associated binary image for esti-
claims be construed as encompassing all the features of claims be construed as encompassing all the features of mation of porosity of the image in FIG. 30A; patentable novelty which reside in the present invention, FIG. 30C illustrates a cross-sectional SEM image of a including all features which would be treated as equivalents modified polyamide imide (e.g., TORLON®) hollow fiber; thereof by those skilled in the art to which the invention FIG. 30D illustrates an associated binary images for $_{20}$ pertains. Therefore, the scope of the present i timation of the porosity of the image in FIG. 30C; defined only by the appended claims, along with the full FIG. 31A illustrates exemplary cross-sectional SEM scope of equivalents to which such claims are entitled.

approach as in Case 1;
 $\frac{25 \text{ or tubular membrane module at a controlled location, and to}}{FIG. 31B.}$ illustrates exemplary cross-sectional SEM subsequently activate the membrane and measure separation FIG. 31B illustrates exemplary cross-sectional SEM subsequently activate the membrane and measure separation
images of a ZIF-8 membrane grown on a polyamide-imide
(e.g., TORLON®) hollow fiber using isothermal IMMP By using membrane growth variables and post-treatments can be controlled accurately. This device also allows in situ char-FIG. 31C illustrates exemplary cross-sectional SEM 30 controlled accurately. This device also allows in situ char-
acterization of the permeation properties of the module after

(e.g., TORLON®) hollow fiber using initial heating IMMP
approach as in Case 3;
FIG. 32A illustrates a chart of Length (mm) vs Thickness
(um) for a ZIF-8 membrane grown on a polyamide-imide
(e.g., TORLON®) hollow fiber usin

hollow fiber or tubular module may have an inner (bore) diameter of about 50 μ m to about 5000 μ m. In an embodi-

FIG. 33 illustrates a chart of Pressure (psi) vs. Flux elliptical base), and the like. In an embodiment, a reactor membrane thickness (mol/m·s), showing helium permeabili- module 200 was fabricated to have a base shape and height, and a reactor module cover 420 was fabricated to

of stainless steel , stainless steel alloys such as MONEL® FIG. 34B illustrates a chart of Temperature (C) vs. Per-
eance (G.P.U.) and Separation Factor, showing ternary attional, Inc.), and the like. In an embodiment, the plastic Co.)), polytetrafluorethylene (PTFE) (e.g., TEFLON® (Du-

Pont Co.)), and the like. In an embodiment, stainless steel 295 extending through the reactor module cover 420 and 304 was used to fabricate the reactor module 200 and cover aligning with the temperature probe diameter 295 304 was used to fabricate the reactor module 200 and cover aligning with the temperature probe diameter 295 of the
420.

and cylindrical-like reactor modules and covers. Accordingly, although the reactor modules and covers. Accordingly, although the rectangular reactor module 200 and cover
ingly, although the rectangular reactor module 200 a

from metal blocks, a person of ordinary skill in the art surface 235, and a fourth hole 280 extending into the
(POSITA) could easily adapt these teaching to other suitable reaction chamber 245 from a fourth side surface 2 (POSITA) could easily adapt these teaching to other suitable is reaction methods of fabricating parts. In an embodiment, the fabrication method may be selected from machining, molding, In an embodiment, the first 265 and second 270 holes may
printing and combinations thereof. For example, if a plastic extend only partially into the first 225 and seco material is used, the reactor module and cover may be surfaces, respectively, of the reactor chamber 245 such that molded by compression or injection molding or printed on $_{20}$ a first and second smaller hole may extend molded by compression or injection molding or printed on $_{20}$ a first and second smaller hole may extend into the reaction a 3-D printer as customary in the art. Accordingly, although chamber 245 to accept a hollow fibe a 3-D printer as customary in the art. Accordingly, although machining is discussed in detail below, this discussion machining is discussed in detail below, this discussion module 100 as shown in FIG. 3. In an embodiment, the should not be interpreted to exclude molding and printing hollow fiber or tubular membrane module 100 may be held should not be interpreted to exclude molding and printing hollow fiber or tubular membrane module 100 may be held
(and sealed) in place in the first and second smaller holes

An upper, right front perspective view of a reactor module 25 with an adhesive such as an epoxy. In an embodiment, the 200 is depicted in FIG. 2. As shown in FIG. 2, the reactor first 265 second 270 third 275 and fourth 28 200 is depicted in FIG. 2. As shown in FIG. 2, the reactor first 265, second 270, third 275 and fourth 280 holes are module 200 has a first length 205, a second length 210 and threaded to accent various fittings as are cu module 200 has a first length 205, a second length 210 and threaded to accept various fittings as are customarily used in a third length 215. Further, the reactor module 200 has an the art. a third length 215. Further, the reactor module 200 has an
upper surface 220, a first side surface 225, a second side
surface 230 (not shown), a third side surface 235 and a
fourth side surface 240 (not shown). The reacto

The reactor module 200 has a reactor chamber 245 with $\frac{35}{25}$ each other to facilitate mixing during flow conditions. a reactor chamber diameter 250 and reactor chamber depth For example, a laboratory-scale reactor module 300 was
255 (not above) attaching into the genetary political 200 from constructed from a rectangular stainless steel 255 (not shown) extending into the reactor module 200 from constructed from a rectangular statiness steel block with a
the unge outflex 220. Earther the reactor module 200 healing first length 305 of about 3-inches, a seco

In an embodiment, the reactor chamber may be cylindri-
cal. In such cylindrical embodiment, the inner diameter of stainless steel block.
the O ring groove 260 is slightly larger than and offset from the Next, first 365, se the O-ring groove 260 is slightly larger than and offset from 45 the reactor chamber diameter 250.

cylindrical. For such non-cylindrical embodiments, and fourth 340 (not shown) side surfaces of the stainless
C ring groove with an inner dimension cliently lerger than O-ring groove with an inner dimension slightly larger than steel block. Although $\frac{1}{2}$ -nich NPT fittings were used, a and affect from an outer dimension of the reactor chember 50. POSITA could easily adapt this teach and offset from an outer dimension of the reactor chamber 50° POSITA could easily adapt this teaching to other fittings.
Accordingly, although the $\frac{1}{6}$ -inch NPT fittings are dis-

In an embodiment, the reactor module 200 has a temperature of the fittings.
ture probe chamber 290 (not shown) with a temperature of the fittings. ture probe chamber 295 (not shown) and a temperature probe The first 365 and second 370 holes extended only partially
probe diameter 295 (not shown) and a temperature probe diameter probe diameter $\frac{325}{25}$ and second

12

420. Although a rectangular reactor module 200 and cover 420 $\frac{420}{\text{The reactor module cover 420 may be constructed from a are depicted in FIGS. 2-4, a person of ordinary skill in the $\frac{5}{\text{metal or a plastic as discussed above. In an embodiment, the}}$$

From a second side surface surface reaction of the surface extending into the reaction chamber 245 from a third side
Although the reactor module and cover were machined extending into the reaction chamber 280 extending in

the sharp in place in the first and second smaller holes
An upper, right front perspective view of a reactor module 25 with an adhesive such as an enoxy. In an embodiment the

constructed from stainless steel 304. may be slightly offset either vertically or horizontally from constructed from stainless steel 304. The meature of the steel 35 each other to facilitate mixing during flow conditions.

the upper surface 220. Further, the reactor module 200 has
an O-ring groove 260 extending into the reactor module 200 has
from the upper surface 220 and surrounding the reactor
chamber diameter 350 of about 2-inches and a about 1.38-inches was machined into the center of the

were drilled for $\frac{1}{8}$ -inch NPT fittings through the first 325 (not shown), second 330 (not shown), third 335 (not shown) In an embodiment, the reactor chamber may not be (not shown), second 330 (not shown), the 330 (not shown) side surfaces of the stainless may be machined in the upper surface of the reactor module. Accordingly, although the 18-inch NPT fittings are dis-
Le on ombodiment the master module 200 has a temporary closed, this disclosure should not be interpreted t

depth (not shown) extending into the reactor module 200
from the upper 220 or first side 225, second side 230, third
side 240, fourth side 245 surface. Suitable temperature
side 240, fourth side 245 surface. Suitable tempe upper surface 225 to attach a reactor module cover (not remained and second smaller holes with a diameter of about shown). In an embodiment, if the temperature probe cham-
 0.02 -inch were drilled through the 0.1-inch wal shown). In an embodiment, if the temperature probe cham-
being 0.02 -inch were drilled through the 0.1-inch wall to create an
ber 290 extends into the reactor module 200 from the upper ϵ_5 aperture for mounting the ho ber 290 extends into the reactor module 200 from the upper 65 aperture for mounting the hollow fiber or tubular membrane surface 220, the reactor module cover 420 will have a module 100. These smaller holes may be used surface 220, the reactor module cover 420 will have a module 100. These smaller holes may be used to support the temperature probe hole with a temperature probe diameter hollow fiber or tubular membrane module 100. The hol hollow fiber or tubular membrane module 100. The hollow

Although the smaller holes were used to support the Atemperature probe chamber 290 with temperature probe hollow fiber or tubular membrane, a person of ordinary skill $\frac{5}{10}$ diameter 295 was machined in the reactor module 300, 400 in the art (POSITA) could easily adapt these teachings to use outside of the O-ring groups 3 in the art (POSITA) could easily adapt these teachings to use outside of the O-ring groove 360 to receive a temperature
an insert to support the hollow fiber or tubular membranes(s) neglected as a bown in FIGS 2.4. As sho or to bundle the hollow theer or tubular membranes such that
the smaller holes are unnecessary. Accordingly, although
smaller holes are discussed in detail above, this discussion
should not be interpreted to exclude other supporting the hollow fiber or tubular membrane. $\frac{\text{peta}}{\text{Area}}$ and the like.

The third 375 and fourth 380 holes extending into the and the like.
In $\frac{1}{2}$ and fourth 340 side surfaces respectively of the $\frac{1}{2}$ A rectangular block was used to fabricate a reactor third 335 and fourth 340 side surfaces, respectively, of the A rectangular block was used to fabricate a reactor reactor module 300 (i.e. surfaces nernendicular to the 15 module cover 420. A photograph of a reactor mo reactor module 300 (i.e., surfaces perpendicular to the mounted hollow fiber or tubular membrane module 100) and cover 420 is depicted in FIG. 4. As shown in FIG. 4, the were offset vertically from each other by about $\frac{1}{2}$ to $\frac{1}{4}$ -inch reactor module cover 420 has a were offset vertically from each other by about $\frac{1}{2}$ to $\frac{1}{4}$ -inch reactor module cover 420 has a first length 425 of about to facilitate mixing during flow conditions. In an embodi-
3-inches, a second length 430 to facilitate mixing during flow conditions. In an embodi-
ment, the third 375 and fourth 380 holes extended into the ₂₀ and a third length 435 (not shown) of about $\frac{1}{4}$ -inch. A ment, the third 375 and fourth 380 holes extended into the $_{20}$ third 335 and fourth 340 sides, respectively, at a height of third 335 and fourth 340 sides, respectively, at a height of plurality of holes (not shown) was machined in the reactor about $1\frac{1}{8}$ -inch such that the third 375 and fourth 380 holes module cover 420 extending through about $1\frac{1}{3}$ -inch such that the third 375 and fourth 380 holes module cover 420 extending through the cover 420 and extended into the reactor chamber 345.

In an embodiment (not shown), the first hole 365 may reactor module 400. In an embodiment, the reactor module extend into the reactor chamber 345 from the first side 25 cover 420 was constructed from stainless steel 304 extend into the reactor chamber 345 from the first side 25 cover 420 was constructed from stainless steel 304.
surface 325, the second hole 370 opposing the first hole 365 A temperature probe hole (not shown) was machined may extend into the reactor chamber 345 from the second the reactor module cover 420 extending through the cover surface 330, the third hole 375 may extend into the reactor 420 and aligning with the temperature probe diame surface 330, the third hole 375 may extend into the reactor 420 and aligning with the temperature probe diameter 295 of chamber 345 from the first surface 325 and the fourth hole the temperature probe chamber 290 in the re chamber 345 from the first surface 325 and the fourth hole the temperature probe chamber 290 in the reactor module 380 opposing the third hole 375 may extend into the reactor $30\,$ 400. As illustrated in FIG. 4, the temp

In an embodiment (not shown), the first hole 365 may probe hole (not shown) of the cover 420 and into the extend into the reactor chamber 345 from the first side temperature probe chamber 290 (not shown) of the reactor extend into the reactor chamber 345 from the first side temperature probe chamber 290 (not shown) of the reactor surface 325, the second hole 370 opposing the first hole 365 \sim module 400. Suitable temperature probes 49 surface 325, the second hole 370 opposing the first hole 365 $\frac{35}{2}$ module 400. Suitable temperature probes 490 include heat may extend into the reactor chamber 345 from the second $\frac{35}{2}$ probes resistance tempera surface 330, the third hole 375 may extend into the reactor chamber 345 from an upper surface of a reactor module chamber 345 from an upper surface of a reactor module The reactor module cover 420 provides a flange-seal by
cover 420 and the fourth hole 380 opposing the third hole identified the cover 420 onto the reactor module 400 vi cover 420 and the fourth hole 380 opposing the third hole tightening the cover 420 onto the reactor module 400 via the 375 extends into the reactor chamber 345 from a bottom Δ_0 plurality of 10-32 hexagonal screws 485.

375 extends into the reactor chamber 345 from a bottom $_{40}$ plurality of 10-32 hexagonal screws 485.

surface of the reactor chamber 345.

Although a few possible alternative configurations for an and the reactor module inlet (i.e., third hole 375) and an outlet (i.e., fourth hole 380) a temperature between about 0° C. and about 200° C. As
for an outer (shell) side solution have been discussed above, shown in FIG 4, the laboratory-scale a POSITA could easily adapt this teaching to other designs. 45 was placed on a stir plate, and heated to about 30° C. with For example, the inlet and the outlet of the outer (shell) side a temperature probe 490. In an e

airtight to be used to measure gas and liquid permeation. To Although a temperature probe 490 was used to heat the provide an airtight seal, an O-ring groove 360 with a 50 reactor module 400 a POSITA could easily adap provide an airtight seal, an O-ring groove 360 with a 50 reactor module 400, a POSITA could easily adapt this diameter of about 2 $\frac{1}{4}$ -inches and cross-section of about reaching to other heating and/or cooling met diameter of about 2 $\frac{1}{4}$ -inches and cross-section of about teaching to other heating and/or cooling methods. Accord-
0.06-inches (AS568-035) was machined in the upper surface in gly, although the temperature probe hea 0.06-inches (AS568-035) was machined in the upper surface ingly, although the temperature probe heating method is 320 of the reactor module 300 to receive an O-ring (see FIG. discussed in detail, this discussion shoul

In an embodiment, the reactor chamber may not be 55 cylindrical. For such non-cylindrical embodiments, an

corner of the upper surface 320 of the reactor module 300, chillers, heat sinks, fans, and the like.

400 about 0.35-inches away from the O-ring groove 360 to The temperature of the reactor module 400 may be set

160 reac receive a plurality of 10-32 hexagonal screws 485 as shown with a temperature probe 490 or measured with a thermomin FIGS. 3-4. Although 10-32 hexagonal screws were used ϵ_5 eter. Although a temperature probe 490 was u in FIGS. **3-4.** Although 10-32 hexagonal screws were used δ eter. Although a temperature probe 490 was used, a POSITA to secure a reactor module cover 420 (see FIG. 4) to the could easily adapt this teaching to other c to secure a reactor module cover 420 (see FIG. 4) to the could easily adapt this teaching to other combinations of reactor module 300, a POSITA could easily adapt this heaters (discussed above) and temperature probes. Suit

fiber or tubular membrane module 100 may be mounted (and teaching to other fasteners. Accordingly, although the 10-32 sealed) in place in the first and second smaller holes with an hexagonal screws are discussed in detail, sealed) in place in the first and second smaller holes with an hexagonal screws are discussed in detail, this discussion adhesive such as an epoxy.

an insert to support the hollow fiber or tubular membranes (s) probe 490 as shown in FIGS. 2-4. As shown in FIG. 4, the or to bundle the hollow fiber or tubular membranes such that temperature probe 490 (e.g., best probe,

tended into the reactor chamber 345.
In an embodiment (not shown), the first hole 365 may reactor module 400. In an embodiment, the reactor module

chamber 345 from the second surface 330. heat probe, thermometer) is inserted in the temperature
In an embodiment (not shown), the first hole 365 may be hole (not shown) of the cover 420 and into the probes, resistance temperature detectors (RTDs), thermo-
couples, thermometers, and the like.

shown in FIG. 4, the laboratory-scale reactor module 400 For example, the inlet and the outlet of the outer (shell) side
solution may be positioned to create turbulent flow.
As depicted in FIG. 4, the reactor module 400 must be temperature probe chamber 290 of the reactor modul

3). In an embodiment, the reactor chamber may not be 55α exclude other heating and/or cooling methods. For example, suitable heaters include temperature probes, hot cylindrical. For such non-cylindrical embodiments, an plates heating coils enclosure heaters silicone rubber heat
O-ring groove with an inner dimension slightly larger than ers polyester heaters polyimide heaters strip hea O-ring groove with an inner dimension slightly larger than ers, polyester heaters, polyimide heaters, strip heaters, band and offset from an outer dimension of the reactor chamber heaters tubular heaters cartridge heaters and offset from an outer dimension of the reactor chamber heaters, tubular heaters, cartridge heaters, and the like. For may be machined in the upper surface of the reactor module. may be machined in the upper surface of the reactor module. $\frac{60}{10}$ example, suitable coolers include liquid baths, cooling coils,
A plurality of threaded holes 385 were machined into each enclosure coolers, thermoele

heaters (discussed above) and temperature probes. Suitable

15
temperature probes 490 include resistance temperature

Mounting/Sealing Hollow Fiber in Reactor Cell/Module

A length of a hollow fiber or tubular membrane module

100 may be threaded through the first and second smaller

holes in the reactor module 200, 300. In an embodiment, Advanced 1 of How module 100 with a reflection of the first and second 10 Reactions that are performed under flowing conditions are assumed 4-inches was threaded through the first and second 10 Reactions that are performed

be mounted (and sealed) in place in the first and second of the hollow fiber 100 (see FIGS. 6-7).

First, an aqueous Zn^{+2} solution containing about 0.22 g

smaller holes with an adhesive such as an enoxy (e.g. smaller holes with an adhesive such as an epoxy (e.g., First, an aqueous $Zn =$ solution containing about 0.22 g
transluscent epoxy (3M)) In an embodiment the hollow ¹⁵ Zinc nitrate hexahydrate (Zn^{+2}) in about 40 mL transluscent epoxy (3M)). In an embodiment, the hollow ¹³ Zinc nitrate hexanydrate ($\text{Zn}^{\prime\prime}$) in about 40 mL deionized
fiber 100 was held (and sealed) in place in the first and water (DI) (about 0.018 mol/L) was f fiber 100 was held (and sealed) in place in the first and water (DI) (about 0.018 mol/L) was flowed through the bore second smaller holes with a small drop of epoxy on the shell 110 at 2 mL/min while an aqueous mIm soluti second smaller holes with a small drop of epoxy on the shell 110 at 2 mL/min while an aqueous mIm solution containing
of each fiber where the fiber passes through the aperture. 9 g mIm in about 80 mL deionized water (DI of each fiber where the fiber passes through the aperture. 9 g mlm in about 80 mL deionized water (DI) (about 1.37
After the epoxy was allowed to cure for about 90 minutes, ₂₀ mol/L) was poured on the outer (shell) side After the epoxy was allowed to cure for about 90 minutes, $_{20}$ the excess fiber ends were removed with tweezers, taking the excess fiber ends were removed with tweezers, taking reaction was stopped after 6 hours and the hollow fiber 100 care not to crush or block the fiber ends. In an embodiment, was rinsed with deionized water (DI) and met a plurality of hollow fibers 100 may be mounted (and sealed) A cross-sectional view of a SEM micrograph of a ZIF-8
membrane grown under agueous flow conditions is shown in

To ensure that the mounted hollow fiber(s) 100 was/were 25 FIG. 6A; and a detailed (10x) cross-sectional view of the properly sealed and that the ends were not crushed or $\sum_{k=1}^{\infty}$ micrograph of the ZIE-8 membrane gr properly sealed and that the ends were not crushed or SEM micrograph of the ZIF-8 membrane grown under blocked, the reactor module 400 should be tested for use as

blacked, the reactor module 400 shown in EUC. 6B. Based upon
a permeation cell by measuring the leak rate and the N₂
permeation cell by measuring the leak rate and the N₂
permeation Cell
Derformance as Permeation Cell

for a hollow fiber or tubular membrane synthesis, a series of membrane grown under 1-octanol flow conditions is shown constant flow membrane synthesis experiments were per-
in FIG. 7A: and a detailed $(10\times)$ cross-sectio constant flow membrane synthesis experiments were per-
formed. Several examples of ZIF-8 membrane fabrication 45 SEM image of the ZIF-8 membrane grown under 1-octanol formed. Several examples of ZIF-8 membrane fabrication 45 SEM image of the ZIF-8 membrane grown under 1-octanol using this reactor module 400 are described below, and their flow conditions is shown in FIG. 7B. Based upon F results are shown in the SEM micrographs of FIGS. 5-8. 7A-7B, the ZIF-8 membrane is formed on the inner (bore)
Performance in Membrane Synthesis: Static Conditions side surface 110 of the hollow fiber 100.
Starting with a

LON®) hollow fiber 100 mounted in the reactor module 50 1-octanol (see FIG. 7) flow systems, the ZIF-8 membrane
400¹² an aqueous synthesis gel consisting of about 0.22 q growth was observed in the inner (bore) side surf 400,¹² an aqueous synthesis gel consisting of about 0.22 g growth was observed in the inner (bore) side surface 110 of $\frac{1}{2}$ in about 40 mL of degree $\frac{1}{2}$ the hollow fiber 100 instead of on the outer (shell) si Zinc nitrate hexahydrate (Zn^{+2}) in about 40 mL of deionized the hollow water (DI) (about 0.018 mol/L) and about 9 g 2-methyl surface 105. midazole (mIm) in about 80 mL deionized water (DI)
(about 1.37 mol/L) was poured on the outer (shell) side
 $\frac{55}{100}$ The next experiment involved using interfacial synthesis

ZIF-8 membrane grown on the outer (shell) side surface 105 the outer (shell) side surface 105 of the hollow fiber 100 . The of the seeded polyamide-imide (e.g., TORLON®) hollow 65 reaction was stopped after 6 hours a of the seeded polyamide-imide (e.g., TORLON®) hollow 65 reaction was stopped after 6 hours and the hollow fiber 100
fiber is shown in FIG. 5B. Based upon FIGS. 5A-5B, the was flushed with neat 1-octanol, heptane, and deion ZIF-8 membrane is formed on the outer (shell) side surface

temperature probes 490 include resistance temperature 105 of the hollow fiber 100. Cross-sectional images of ZIF-8 detectors (RTDs), thermocouples, thermometers, and the inner surface membrane were examined using Hitachi S detectors (RTDs), thermocouples, thermometers, and the inner surface membrane were examined using Hitachi SU
16 8010 scanning electron microscope after a thin layer of gold

The hollow fiber or tubular membrane module 100 may branes were grown by flowing reagents through the bore 110 membrane module 100 may branes were grown by flowing reagents through the bore 110 membrane module 100 (see

ing this same method.
To ensure that the mounted hollow fiber(s) 100 was/were 25 FIG 6A; and a detailed (10x) cross-sectional view of the

(about 1.37 mol/L) was poured on the outer (shell) side ⁵⁵
surface 105 of the hollow fiber 100. After 6 hours at 30° C.
the shell solution was removed and the hollow fiber 100 was
to physically separate the organic and A cross-sectional view of a SEM micrograph of a ZIF-8
membrane grown on an outer (shell) side surface of a seeded
polyamide-imide (e.g., TORLON®) hollow fiber is shown
in a alueous mlm solution containing about 9
polyamid was flushed with neat 1-octanol, heptane, and deionized water (DI).

To determine the effect of flow rate, the aforementioned Second, a $Zn^{+2}/1$ -octanol solution containing about 0.22
experiment was repeated using a slower flow rate $(1 \alpha \text{ z})$ α and $Zn^{+2}/1$ -octanol solution containi experiment was repeated using a slower flow rate $(1 \text{ g Zinc nitrate hexahydrate } (2n^{+2}) \text{ in about 40 mL 1-octanol}$
(about 0.018 mol/L) was held static in the bore 110 while an

A cross-sectional view of a SEM micrograph of a ZIF-8 aqueous mIm solution containing about 9 g mIm in about 80
membrane grown under 1-octanol flow conditions of 2⁵ mL dejonized water (DI) (about 1.37 mol/L) was poured membrane grown under 1-octanol flow conditions of 2 \degree mL deionized water (DI) (about 1.37 mol/L) was poured on mL/min bore flow rate, using an interfacial system is shown

permeation properties of three of the ZIF-8 membranes however, the membrane had a low H_2/C_3H_8 selectivity.

weing the reactor module 400. The entire process of mem. A chart of Time vs. Flow Rate for pulsed flow condit using the reactor module 400. The entire process of mem-
here are a chart of Time vs. Flow Rate for pulsed flow conditions
here $\frac{10}{10}$ of the hollow fiber 100 is illustrated in brane formation and permeation measurement can be done 20 and 200 and \overline{F} FIG. 12. Under pulsed flow conditions, a ZIF-8 membrane in situ within the same pre-fabricated reactor module 400,
thereby expecting the difficulties associated with apprions was formed on the inside of the bore 110 of the hollow thereby avoiding the difficulties associated with previous was formed on the inside of the bore 110 of the hollow
membrane 100, and, further, the pulsed-flow membrane had

Measurement of Single-Component Gas Permeation Properties for In-Situ grown ZIF-8 Membrane							
		Flow	Thick- ness	Permeance (G.P.U.)		Selectivity	30
Example	Support	Rate	(μm)	N,	SF_{6}	N_2/SF_6	
ZIF-8 Outer (see FIGS. $5A-5B$	Torlon Tube	Static	3	800	430	1.9	35
ZIF-8 Inner 1 (see FIG. $8A$)	Torlon Tube	2 mL min	$5 - 15$	240	100	2.4	
$ZIF-8$ Inner 2 (see FIG. $8B$)	Torlon Tube	$1 \mu L$ mın	$2 - 3$	320	60	5.3	

 40 confirming the structure of supported ZIF-8 membranes is growth step. The reaction was stopped after about 9 hours shown in FIG. 9. XRD patterns were measured on a PANa- and the hollow fiber 100 was rinsed with deionized w shown in FIG. 9. XRD patterns were measured on a PANa- and the hollow fiber 100 was rinsed with deionized water lytical X'Pert Pro diffractometer at room temperature using 45 (DI) and methanol. Cu Kα radiation of λ=0.154 nm and a scanning range of Second, a Zn⁺²/1-octanol solution containing about 0.022
5-40° 20.
9 Zinc nitrate hexahydrate (Zn⁺²) in about 40 mL 1-octanol

An EDS Zinc line profile scan of distance from membrane (about 0.018 mol/L) was flowed through the bore 110 at surface (μ m (microns)) vs. intensity, depicting the thickness about 10 μ L/hour while a mIm/1-octanol sol of the ZIF-8 membrane to be about 3 μ m is shown in FIG. ⁵⁰ 10.

flowing reagents through the bore 110 of the hollow fiber ⁵⁵ After about 20 minutes, the pump was stopped for another 100 (see FIGS. 6-7, 18B-18C) under static, continuous and 3.5 hours to provide another static growth s 100 (see FIGS. 6-7, 18B-18C) under static, continuous and 3.5 hours to provide another static growth step. The reaction pulsed flow conditions. See FIGS. 11-12 & 18A. was stopped after about 9 hours and the hollow fiber 10

First, a $\text{Zn}^{+2}/1$ -octanol solution containing about 0.22 g rinsed with 1-octanol, deionized water (DI), heptanes and Zinc nitrate hexahydrate (Zn^{+2}) in about 40 mL 1-octanol _{co} methanol. Zinc intrate hexahydrate (Zn^2) in about 40 mL 1-octanol $\frac{60}{12}$ methanol.

(about 0.018 mol/L) was continuously flowed through the Third, a $Zn^{+2}/1$ -octanol solution containing about 0.11 g

bore 110 at about 60 µL containing about 9 g mIm in about 80 mL deionized water (about 0.018 mol/L) was flowed through the bore 110 at (DI) (about 1.37 mol/L) was poured on the outer (shell) side about 10 μ L/hour while an aqueous mIm solution (DI) (about 1.37 mol/L) was poured on the outer (shell) side about 10 μ L/hour while an aqueous mIm solution containing surface 105. The reaction was stopped after about 9 hours 65 about 9 g mIm in about 80 mL deioniz surface 105. The reaction was stopped after about 9 hours 65 about 9 g mlm in about 80 mL deionized water (DI) (about and the hollow fiber 100 was rinsed with 1-octanol, hep- 1.37 mol/L) was poured on the outer (shell) tanes, deionized water (DI) and methanol. After about 2 hours of bore solution flow, the pump was

18

 μ min).
A cross-sectional view of a SEM micrograph of a ZIF-8 agree and solution containing about 9 g mIm in about 80

mL/min bore flow rate, using an interfacial system is shown
in FIG. 8A; and a cross-sectional view of the SEM micro-
graph of the ZIF-8 membrane grown under 1-octanol flow
conditions of 1 µL/min bore flow rate, using the

membrane fabrication processes.

a higher H_2/C_3H_8 selectivity than the continuous-flow mem-
 25 brane discussed above.

TABLE 1

²⁵ brane discussed above.

²⁵ brance discussed above .

To further test the effect of flow conditions, the afore-

mentioned experiment was repeated using a one phase aqueous solvent system, a single phase organic solvent system (1-octanol), and a biphasic interfacial solvent system

(aqueous/1-octanol), respectively.
First, an aqueous Zn^{+2} solution containing about 0.22 g
Zinc nitrate hexahydrate (Zn^{+2}) in about 40 mL deionized water (DI) (about 0.018 mol/L) was flowed through the bore 110 at about 10 μ L/hour while an aqueous mlm solution containing about 9 g mlm in about 80 mL deionized water (DI) (about 1.37 mol/ L) was poured on the outer (shell) side surface 105. After about 2 hours of bore solution flow, the pump was stopped for about 3.5 hours to provide a static growth step. After about 3.5 hours, the pump was continued An X-ray Diffraction (XRD) chart of 2Theta (\degree) vs. \degree for about 20 minutes. After about 20 minutes, the pump was intensity for simulated and experimental ZIF-8 membranes, stopped for another 3.5 hours to provide anoth stopped for another 3.5 hours to provide another static

 540° 20. g Zinc nitrate hexahydrate (Zn^{+2}) in about 40 mL 1-octanol
An EDS Zinc line profile scan of distance from membrane (about 0.018 mol/L) was flowed through the bore 110 at about 10 μ L/hour while a mIm/1-octanol solution containing about 9 g mIm in about 80 mL 1-octanol (about 1.37 mol/L) 10. was poured on the outer (shell) side surface 105. After about
Performance in Membrane Synthesis: Static, Continuous 2 hours of bore solution flow, the pump was stopped for Performance in Membrane Synthesis: Static, Continuous 2 hours of bore solution flow, the pump was stopped for and Pulsed Flow Conditions $\frac{1}{2}$ hours to provide a static growth step. After about d Pulsed Flow Conditions about 3.5 hours to provide a static growth step. After about 1 an embodiment, ZIF-8 membranes were grown by 3.5 hours, the pump was continued for about 20 minutes. ilsed flow conditions. See FIGS. 11-12 & 18A. was stopped after about 9 hours and the hollow fiber 100 was First, a $Zn^{2/1}$ -octanol solution containing about 0.22 g rinsed with 1-octanol, deionized water (DI), heptanes

stopped for about 3.5 hours to provide a static growth step. fiber 100. The flow profile included an initial continuous After about 3.5 hours, the pump was continued for about 20 growth step, followed by a static growth st

static, continuous and pulsed flow conditions through the bore 110 of the hollow fiber 100 is illustrated in FIG. 18A.

A cross-sectional view of a SEM micrograph of a ZIF-8
membrane grown in a polyamide-imide (e.g., TORLON®)
hollow fiber with a static flow of a $\text{Zn}^{+2}/1$ -octanol bore
colution is illustrated in EIG 18P. As shown in EIG nonow moet 100 to sustain the film growth after the initial 20
nucleation communed the ZIT-8 crystal subclue of the film.
side surface 110 of the hollow fiber 100.
Performance in Membrane Synthesis: Pulsed Flow Synthe-

A cross-sectional view of a SEM micrograph of a ZIF-8 membrane grown on a polyamide-imide (e.g., TORLON®) Table 2 shows measurement of gas and liquid permeation hollow fiber with continuous flow of a $Zn^{2}/1$ -octanol bore properties for various supported ZIF-8 membranes. hollow fiber with continuous flow of a $\text{Zn}^{+2}/\text{1}$ -octanol bore

After about 3.5 hours, the pump was continued for about 20
minutes. After about 20 minutes, the pump was stopped for
about 20 minutes. After about 20 minutes, the pump was stopped for
about a brief reactant replenishment 2 hours, followed by a static growth step of about 0 μ L/hour of the bore solution for about 3.5 hours, followed by a A chart of Time vs. Flow Rate summarizing the examined of the bore solution for about 3.5 hours, followed by a
Actic, continuous and pulsed flow conditions through the replenishment step of about 10 µL/hour of bore solutio bore 110 of the hollow fiber 100 is illustrated in FIG . 18A . " about 20 minutes and followed by another static growth step

solution is illustrated in FIG. 18B. As shown in FIG. 18B, $\frac{\text{Aquation}}{15}$ images of ZIF-8 membranes formed on two hollow fibers static growth conditions produce dense, non-continuous ¹⁵ images of ZIF-8 membranes formed on two hollow fibers coatings of ZIF-8 particles in the inner (bore) side surface 100; and FIG. 17B shows the zinc elemental mapp 110 of the hollow fiber 100. This is due to the lack of confirming the localization of the membrane to the inner
sufficient $7n^{2+}$ ions available in the microscale bore of the (bore) side surface 110 of the hollow fiber sufficient Zn^{2+} ions available in the microscale bore of the (bore) side surface 110 of the hollow fiber 100. X-ray hollow fiber 100 to sustain the film growth after the initial α diffraction confirmed the ZIF-

sis

TABLE 2

Measurement of Gas and Liquid Permeation Properties									
		Thickness	Permeance (G.P.U.)						
Reference	Support	(num)	Н,				CO_2 CH_4 C_3H_8 C_6H_{14} C_6H_{12}		C_6H_6
Y. Pan Y. Pan AJB	YSZ Tube $\mathrm{Al}_2\mathrm{O}_2$ Disk Torlon ®	2.5 2.5 8	4400 1100 2900	1200 390 500	360 240 270	4 2 60	2600	600	290

solution is illustrated in FIG. 18B. As shown in FIG. 18C, continuous flow growth conditions produced a thin, continuous membrane (about 3 μ m thickness) that was formed $\frac{40}{\sigma}$ in the inner (bore) side surface 110 of the hollow fiber 100 . This is due to the relatively rapid transport of reactants to the interface under continuous flow, leading to rapid formation and closure of the ZIF-8 layer. The growing membrane itself becomes a barrier between the two immiscible solvents, and confines the liquid-liquid interface into the gaps and interstices between the ZIF-8 crystals. See FIG. 15.
A cross-sectional view of a SEM micrograph of a ZIF-8

membrane grown on a polyamide-imide (e.g., TORLON®) hollow fiber with intermittent flow of a $\text{Zn}^{\times}/\text{I}$ -octanol bore solution is illustrated in FIG. 18D. As shown in FIG. 18D,

A X-ray Diffraction (XRD) chart of 2Theta $(°)$ vs. intensity for simulated and experimental ZIF-8 membranes and polyamide-imide (e.g., TORLON®) hollow fiber, confirming formation and structure of supported ZIF-8 membranes
on the bore 110 of the hollow fiber 100 is shown in FIG. 14. XRD patterns were measured on a PANalytical X'Pert Pro diffractometer at room temperature using Cu K α radiation of λ =0.154 nm and a scanning range of 5-40° 20.

Performance in Membrane Synthesis: Pulsed Flow and Stirred Synthesis

Table 3 shows the single-component gas permeation properties of a ZIF-8 membrane measured in situ using the reactor module 400 and gently stirring the outer (shell) side 105 solution.

intermittent flow growth conditions produced a thicker, 65 continuous membrane (about $8 \mu m$ thickness) that was formed in the inner (bore) side surface 110 of the hollow

Table 4 shows measurement of gas permeation properties of a ZIF-8 membrane when the mixture feed consisted of a 1:1 H_2/C_3H_8 mixture.

Using a macroporous polyamide-imide (e.g., TOR- 15 be stirred at about 40 rpm to about 80 rpm.
NON bollow fiber and the material ZIE 8 as an archatyne After about 2 hours of bore solution flow, the pump was LON®) hollow fiber and the material ZIF-8 as an archetype After about 2 hours of bore solution flow, the pump was
for a hollow fiber or tubular membrane synthesis, a series of stopped for about 3.5 hours to provide a stati pulsed flow membrane synthesis experiments were per-
formed Table 5 shows combinations of recursor solutions
formed Table 5 shows combinations of recursor solutions formed. Table 5 shows combinations of precursor solutions minutes. After about 20 minutes, the pump was stopped for
and locations (bore/shall) tested to synthesize $71F-8$ mam_n 20 another 3.5 hours to provide another sta and locations (bore/shell) tested to synthesize ZIF-8 mem- $\frac{20}{2}$ another 3.5 hours to provide another static growth step. In an embodiment, the bore solution flows at a first flow rate for branes using a reactor module 400. Several examples of embodiment, the bore solution flows at a first flow rate for a second $ZIE-R$ membrane fabrication using the reactor module 400 a first period, the bore solution is stop ZIF-8 membrane fabrication using the reactor module 400 a first period, the bore solution is stopped for a second
are described below and their results are shown in the SFM period, the bore solution flows at a second fl are described below, and their results are shown in the SEM period, the bore solution flows at a second flow rate for a fourth micrographs of $FIGS$ 17A-17B 18A-18D and 20A-20D third period and the bore solution is stopped micrographs of FIGS. 17A-17B, 18A-18D and 20A-20D. third period and the bore solution is stopped for a fourth Γ For the experiments, a temperature probe 490 set at 30 $^{\circ}$ C. ²⁵ period. In an embodiment, the first and second flow rate is about 100 μ L/hour. In an embodiment,

Combinations of Precursor Solutions Tested to Synthesize ZIF-8 Membranes Using IMMP Approach						To stop the reaction, about 10 mL of neat 1-octanol solvent was pushed through the bore 110 while the outer
Example	Bore Solution	Shell Solution	Molar Ratio $\text{Zn}^{+2}/\text{mIm}$	Membrane Location		(shell) side surface 105 was soaked in about 70 mL of neat DI (and replaced three times) to remove the excess Zn^{2} . Next, about 10 mL heptanes were pushed through the bore
(see FIGS. 17A- 17B and 18A- 18D)	0.018 mol/L Zn^{+2} in 1- octanol	1.37 mol/L mIm in DI	75	Bore (inner surface)		110 while the outer (shell) side surface 105 was soaked in about 70 mL heptanes to remove the 1-octanol. Next, about 10 mL of hexane was pushed through the bore 110 while the outer (shell) side surface 105 was soaked in hexane to
\overline{c} (see FIG. $20A$)	0.018 mol/L Zn^{+2} in 1- octanol	1.37 mol/L mIm in DI	75	In fiber, closer to 40 bore surface		remove the heptanes. Then, about 20 mL methanol was flowed through the bore 110 while the outer (shell) side
3 (see FIG. 20B)	0.018 mol/L Zn^{+2} in 1- octanol	1.37 mol/L mIm in DI	75	In fiber, closer to bore surface		surface 105 was soaked in about 70 mL methanol to remove the DI. The reactor module 400 was allowed to air dry at least 4 days before permeation testing.
4 (see FIG. $20C$)	1.37 mol/L mIm in DI	0.018 mol/L Zn^{+2} in 1- octanol	75	Shell (outer surface)	45	Example 2
5 (see FIG. $20D$)	1.37 mol/L mIm in DI	0.018 mol/L Zn^{+2} in 1- octanol	75	Shell (outer surface)		First, about 10 mL DI was first flowed through the bore 110 followed by about 3 mL of an aqueous Zn^{+2} (bore)

ing 0.11 g Zinc nitrate hexahydrate (Zn^{2}) in about 40 mL solution flow, the pump was stopped for about 3.5 hours to 1-octanol (about 0.018 mol/L) was flowed through the bore provide a static growth step. After about 3.5 110 of a horizontal hollow fiber 100 at a flow rate of about

10 was continued for about 20 minutes. After about 20 minutes,

10 µL/hour. In an embodiment, a limited $Zn^{+2}/1$ -octanol 60 the pump was stopped for another 3 about 0.03 mol/L Zn^{+2} in 1-octanol may be used. Increasing the Zn^{+2} concentration to about 0.03 mol/L leads to reduc- 65 the Zn^{+2} concentration to about 0.03 mol/L leads to reduc- 65 to remove the excess Zn^{+2} . Then, about 20 mL methanol was tion of the membrane thickness and increase in crystal flowed through the bore 110 while the tion of the membrane thickness and increase in crystal flowed through the bore 110 while the outer (shell) side nucleation.
surface 105 was soaked in about 70 mL methanol to remove

TABLE 4 About 70 mL of an aqueous mlm solution containing about 9 g mIm in about 80 mL dionized water (DI) (about 1.37 mol/L) was slowly poured into the reactor module 400,
immersing the outer (shell) side surface 105 of the hollow
5 ther 100 Jp an embodiment an excess agueous mJm fiber 100. In an embodiment, an excess aqueous mlm solution containing about 0.5 mol/L to about 10 mol/L mlm in deionized water (DI) may be used, provided that the mIm concentration is in excess. In an embodiment, an excess aqueous mIm solution containing about 1.2 mol/L to about

zIF-

8 2250 8 26.4 85 10 equences in model of containing action 1.2 model of the

8 2250 26.4 85 10 equences mim solution was gently street.

The aqueous mim solution containing act (DI) may be used.

The aqueous mim (sh about 60 rpm to prevent the formation of local concentration Pulsed-Flow Synthesis of ZIF-8 Membranes gradients. In an embodiment, the aqueous mIm solution may
Lising a macroporous polyamide imide (e.g. TOP 15 be stirred at about 40 rpm to about 80 rpm.

was inserted into the reactor module 400 during membrane
growth.
growth.
growth.
growth. TABLE 5 30 about 10 minutes to about 30 minutes and the fourth period
30 is about 3 hours to about 4 hours.
To stop the reaction, about 10 mL of neat 1-octanol

First, about 10 mL DI was first flowed through the bore 110 followed by about 3 mL of an aqueous Zn^{+2} (bore) so solution (about 0.018 mol/L). About 70 mL of an aqueous mIm solution (about 1.37 mol/L) was added to the outer Example 1 (shell) side surface 105 (i.e., slowly poured into the reactor module 400 and was gently stirred at about 60 rpm to prevent the formation of local concentration gradients) while First, about 10 mL of neat 1-octanol solvent was flowed prevent the formation of local concentration gradients) while
through a bore 110 using a syringe pump.
Second, about 3 mL of a $\text{Zn}^{+2}/1$ -octanol solution contain

surface 105 was soaked in about 70 mL methanol to remove

the DI. The reactor cell 400 was allowed to air dry at least Second, about 70 mL of a $\text{Zn}^{+2}/1$ -octanol solution (about 4 days before permeation testing. 0.018 mol/L) was added to the outer (shell) side surface 105

First, about 10 mL 1-octanol was first flowed through the
bore 110 followed by about 3 mL of a Zn⁺²/1-octanol (bore)
bore 110 followed by about 3 mL of a Zn⁺²/1-octanol (bore)
solution (about 0.018 mol/L). About 70 mL

surface 105 was soaked in about 70 mL of neat 1-octanol 20 methanol was nowed through the bore 110 while the outer
solvent. Then, about 10 mL of heptanes were pushed (shell) side surface 105 was soaked in about 70 mL of solvent. Then, about 10 mL of heptanes were pushed $($shell)$ side surface 105 methanol. was soaked in about 70 mL of heptanes to remove the Capping of Hollow Fiber Ends
1-octanol. Then, about 10 mL of hexane was pushed through A capping solution contain the bore 110 while the outer (shell) side surface 105 was soaked in about 70 mL of hexane to remove the heptanes. soaked in about 70 mL of hexane to remove the heptanes. may be used as a filler material to cap both ends of the Then, about 10 mL of methanol was pushed through the bore hollow fiber 100. In an embodiment, a capping solut Then, about 10 mL of methanol was pushed through the bore hollow fiber 100. In an embodiment, a capping solution 110 while the outer (shell) side surface 105 was soaked in containing about 9 wt % PDMS in heptane was heated about 70 mL of methanol. The reactor module 400 was allowed to air dry at least 4 days before permeation testing.

0.018 mol/L) was added to the outer (shell) side surface 105 inner (bore) side 110 and followed by curing at about 120°
while the plm/estenal (bore) solution was flowed at about ... C. for about 2 hours. while the mIm/octanol (bore) solution was flowed at about $\frac{1}{40}$ C. for about 2 hours.
10 uJ /hour for about 2 hours After about 2 hours of bore Performance of ZIF-8 Membranes Made Using IMMP Reac-10 μ L/hour for about 2 hours. After about 2 hours of bore μ Performance of μ - 8 Membranes Made Using Immediates Made Using Immediates Made Using Immediates Made Using Immediates Made USING Reaction Figures Made solution flow, the pump was stopped for about 3.5 hours to tor Cell/Module
provide a static growth step. After about 3.5 hours, the pump The separation properties of the ZIF-8 membrane grown provide a static growth step. After about 3.5 hours, the pump The separation properties of the ZIF-8 membrane grown
was continued for about 20 minutes. After about 20 minutes, on the inner bore of the polyamide-imide (e.g. was continued for about 20 minutes. After about 20 minutes, on the inner bore of the polyamide-imide (e.g., TORLON®)
the pump was stopped for another 3.5 hours to provide 45 hollow fiber 100 were characterized by hydrogen

(shell) side surface 105 was soaked in about 70 mL of neat permeation cell. See FIG. 16. In an embodiment, the feed DI (and replaced three times) to remove the excess Zn^{2} . ⁵⁰ mixture may be selected from the group c DI (and replaced three times) to remove the excess Zn^{+2} . ⁵⁰ Next, about 10 mL heptanes were pushed through the bore Next, about 10 mL heptanes were pushed through the bore hydrogen/hydrocarbons, hydrogen/propane, propylene/pro-
110 while the outer (shell) side surface 105 was soaked in anne and butenes/butanes 110 while the outer (shell) side surface 105 was soaked in

about 70 mL heptanes to remove the 1-octanol. Next, about

10 mL of hexane was pushed through the bore 110 while the
 $\frac{1}{55}$ test binary H_2/C_3H_8 and C_3

First, about 10 mL DI was flowed through the bore 110 $_{65}$ samples.
followed by about 3 mL of an aqueous mIm solution (about A gas chromatograph with TCD and FID detectors was used to determine the composition of permea

0.018 mol/L) was added to the outer (shell) side surface 105 while the aqueous mIm (bore) solution was flowed at about Example 3 10 µL/hour for about 2 hours. After about 2 hours of bore solution flow, the pump was stopped for about 3.5 hours to

To stop the reaction, about 10 mL neat 1-octanol solvent
was pushed through the bore 110 while the outer (shell) side
was soaked in about 70 mL of hexane. Then, about 20 mL
surface 105 was soaked in about 70 mL of hexane.

1 - A capping solution containing poly(dimethylsiloxane) 25 (PDMS) (e.g., SYLGARD® 184 (Corning)) in a solvent containing about 9 wt % PDMS in heptane was heated at about 90 $^{\circ}$ C. with vigorous stirring for about 4 hours to thermally crosslink the PDMS. In an embodiment, a capping

Example 4

After cooling to about 25° C., about 2 µL droplet was

drough the bore 110 followed by about 3 mL of

other static growth step.

To stop the reaction, about 10 mL of neat 1-octanol binary equimolar mixture permeation as a function of tem-To stop the reaction, about 10 mL of neat 1-octanol binary equimolar mixture permeation as a function of tem-
solvent was pushed through the bore 110 while the outer perature, with the reactor module 400 directly acting as perature, with the reactor module 400 directly acting as a

remove the heptanes. Then, about 20 mL methanol was
flowed through the bore 110 while the outer (shell) side
surface 105 was soaked in about 70 mL methanol to remove
the DI. The reactor module 400 was allowed to air dry at Example 5 21B, 22A-22B and 23A-23B, the error bars were estimated from characterization of three independent ZIF-8 membrane samples.

used to determine the composition of permeate. At least 3

were reported) at each temperature after waiting 30 minutes for steady-state conditions.

LON®) hollow fibers 100 were formed using the reaction $\frac{3}{2}$ described earlier. The ends of the hollow libers conditions of Example 1 under intermittent flow conditions $\frac{1}{2}$ C₃H₈ (see FIG. 18A: flow profile 2). The as-made ZIF-8 membranes of DAM and C₃H₈ (see FIG. 18A: flow profile 2). The as-made ZIF-8 membranes of polyamide-imide (e.g., TORLON®) hollow fibers c₃H₈ separation beh 125 at 120° C. (see FIG. 22A), and strong temperature
dependence of H_2 permeance with temperature, indicating
activated molecular transport through the ZIF-8 pores. Fur-
ion). With further improvements in the caping me activated molecular transport through the ZIF-8 pores. Fur-
tion. With further improvements in the capping methods
ther, the example ZIF-8 membrane on polyamide-imide $_{15}$ (e.g., using less permeable polymers than PDMS) ther, the example ZIF-8 membrane on polyamide-imide $_{15}$ (e.g., using less permeable polymers than PDMS) and opti-
(e.g., TORLON®) hollow fiber 100 proved to be robust mization of the ZIF-8 membrane growth conditions, t (e.g., TORLON®) hollow fiber 100 proved to be robust mization of the ZIF-8 membrane growth conditions, the showing no decline in permeation properties after six weeks small C_3H_8 permeance can be further reduced and th

ous studies^{8, 17-20} and prevented a high C_3H_6/C_3H_8 separa-
tion factor. See FIG. 23B. It was hypothesized that the high of current processes and moves significantly towards real- C_3 isomer permeances were due to both molecular transport izing scalable molecular sieving MOF membranes.
through the ZIF-8 membrane as well as due to bypassing of 25 Improved Membrane Synthesis Using IMMP Reactor Cell the ZIF-8 membrane by the feed molecules through the ends of the fiber. See FIG. 24A.

To suppress this membrane bypass, the inventors include a capping step to the IMMP approach, accomplished by applying a controlled amount of a solution containing poly 30 (dimethylsiloxane) (PDMS) to the ends of the mounted performed. Several examples of ZIF-8 membrane fabrica-
hollow fibers 100. See FIG. 24B. The PDMS solution is tion using this reactor module 400 are described below, and hollow fibers 100. See FIG. 24B. The PDMS solution is tion using this reactor module 400 are described below, and readily absorbed by capillary action into the hollow fiber their results are shown in SEM micrographs of FIG 100, and blocks the pores of the hollow fiber 100 support. 27A-27F, 30C and 31A-31C.
See FIGS. 25A and 25B. SEM cross-sectional imaging (see 35 FIG. 26A shows a schematic of the IMMP reactor module FIG. 25A) and EDX mappin FIG. 25A) and EDX mapping (see FIG. 25B) indicated that the fiber matrix is completely covered by PDMS while the the fiber matrix is completely covered by PDMS while the surface 110 of the polyamide imide (e.g., TORLON®) bore remains unblocked. Since the permeances of the feed hollow fibers. The formation of a ZIF-8 membrane is bore remains unblocked. Since the permeances of the feed hollow fibers. The formation of a ZIF-8 membrane is gases through PDMS are 3 orders of magnitude lower than achieved by contacting the two reactant streams (i.e., through the macroporous hollow fiber 100 support, the C_3H_8 40 flux should decrease substantially after capping. After curing 110, and 2-methylimidazole (mIm) dissolved in deionized the PDMS-sealed hollow fibers 100, the H_2/C_3H_8 separation water (DI) on the outer (shell) side sur the PDMS-sealed hollow fibers 100, the H_2/C_3H_8 separation water (DI) on the outer (shell) side surface 105) across the factor is now much higher (370 at 120° C.) (see FIG. 21A) polyamide-imide (e.g., TORLON®) hollow f factor is now much higher (370 at 120° C.) (see FIG. 21A) and the C_3H_6/C_3H_8 separation factor is also higher (12 at 25° C .) (see FIG. 21B), consistent with previously reported 45 cible, thereby allowing a sharp interface between the two $ZIF-8$ membranes with low defect densities." $17-20$ Notably, phases. The use of a large excess of mlm (e.g., mlm/ $Zn⁺²$ the C_3H_8 permeance decreased by a factor of 10 after
concentration ratio=75) allows the Zn^{2+} to act as a limiting
capping. This indicates that most of the propane was pre-
visually bypassing the ZIF-8 layer and th capping step to the IMMP largely shuts down this non-50 The temporal flow profile in the bore side is shown in FIG.
selective permeation path. The permeate stream contains 26B. It comprises four steps: about 120 minutes o 92% $C_3H_6/8% C_3H_8$, which is a significant upgrade from the equimolar feed stream.

Performance of Bundle of ZIF-8 Membranes Made Using minutes of membrane growth under static conditions (Step 1955)
IMMP Reactor Cell/Module 55 2), about 20 minutes of Zn^{2±}/octanol solution replenishment

achieved by bundling large numbers of fibers in the perme-
ation module. The present invention has the advantage of 26B, as discussed further below. They comprise an isother-
being inherently a modular approach that shoul being inherently a modular approach that should allow 60 independent processing of membranes in each fiber constiindependent processing of membranes in each fiber consti-
titles and decreased from 42° C. to 30° C. in
about 25 minutes and decreased from 42° C. to 30° C. in

module 400 to the simultaneous processing of three poly-
amide-imide (e.g., TORLON®) hollow fiber 100 supports. 65 ule/Cell amide-imide (e.g., TORLON®) hollow fiber 100 supports. 65 ule/Cell
The processing conditions were identical to the case of the To investigate the membrane formation process, the The processing conditions were identical to the case of the single-fiber membranes, except that the total feed solution

GC injections were collected (median permeance values initially introduced to the bore and the flow rate on the bore were renorted) at each temperature after waiting 30 minutes side was increased by a factor of 3 so that t through individual hollow fibers 100 was maintained constant in relation to the single-fiber membrane fabrication The ZIF-8 membranes in polyamide-imide (e.g., TOR-
N/D) hollow fibers 100 were formed using the reastion 5 described earlier. The ends of the hollow fibers 100 were

of testing and multiple heating/cooling cycles. separation factor further increased, leading to a highly
While the permeation properties were dominated by
molecular sieving, the C_3 isomer (especially C_3H_8) per- 20 range of hydrocarbon separations, the membrane processing of current processes and moves significantly towards realizing scalable molecular sieving MOF membranes.

Using a macroporous polyamide (e.g., TORLON®) hollow fiber and the material ZIF-8 as an archetype for hollow fiber and tubular membrane synthesis, a series of isothermal and initial heating membrane synthesis experiments were

achieved by contacting the two reactant streams (i.e., Zn^{2+} ions dissolved in 1-octanol on the inner bore side surface these experiments, the two solvents are chosen to be immiscible, thereby allowing a sharp interface between the two

equimolar feed stream.

Performance of Bundle of ZIF-8 Membranes Made Using minutes of membrane growth under static conditions (Step
 $\mu L/h$ our of a Zn^{2±}/octanol solution (Step 1), about 210 IMP Reactor Cell/Module 55 2), about 20 minutes of Zn^{2+} /octanol solution replenishment
In large-scale gas separations with hollow fiber mem-
at a continuous flow of about 62 uL/hour (Step 3), and a final In large-scale gas separations with hollow fiber mem-
branes, high membrane areas per unit module volume are
210 minutes of static membrane growth (Step 4).

the sum of the inventors applied reactor about 25 minutes and decreased from 42° C. to 30° C. in To demonstrate this concept, the inventors applied reactor about 60 minutes), as discussed further below.

inventors repeated the IMMP approach multiple times and

arrested the ZIF-8 membrane growth at three different points as indicated by the numbered temporal locations (i.e., arrest as indicated by the numbered temporal locations (i.e., arrest growth step (arrest point $#1$), a dual-layer ZIF-8 membrane may be observed on the inner bore surface 110 of the hollow

low fiber mounted in the IMMP reactor module 400,¹² a ⁵ about 7 μ m (see Table 6), of which about 4.5 μ m is a dense $Zn^{+2}/1$ -octanol solution containing about 0.22 g Zinc nitrate ZIF-8 membrane layer and the rem $Zn^{+2}/1$ -octanol solution containing about 0.22 g Zinc nitrate ZIF-8 membrane layer and the remainder is a discontinuous hexahydrate (Zn^{+2}) in about 40 mL 1-octanol (about 0.018 membrane layer composed of plate-like st mol/L) was flowed through the bore 110 of the hollow fiber
10 Without wishing to be bound to any theory, the inventors
100 while an aqueous mIm solution containing about 9 g of $\frac{1}{10}$ believe this morphology arises fr mlm in about 80 mL deionized water (DI) (about 1.37 mlm (which is soluble in both water and octanol) into the mol/L) was poured into the reactor chamber 345 immersing octanol phase. At the inner surface of the fiber, mlm mol/L) was poured into the reactor chamber 345 immersing octanol phase. At the inner surface of the fiber, mIm may be the outer (shell) side surface 105 of the hollow fiber 100 rapidly consumed by reaction with Zn^{2+} t

The temperature of the reactor module 400 was main-
the mIm species further into the octanol phase, leading to the
tained at about $31.0\pm0.5^{\circ}$ C. throughout the IMMP process.
diffusion-limited formation of a second di

growth step, the flow rate of about 0.6 mL/hour through the replenishment of the limiting reactant Zn^2 , a single layer
bere 110 of the hollow fiber 100 was stormed for about 210 $ZlF-8$ membrane on the inner bore surfa bore 110 of the hollow fiber 100 was stopped for about 210 ZIF-8 membrane on the inner bore surface 110 of the hollow
fiber 100 was observed with an overall thickness of about 10

solvent was pushed through the bore 110 while the outer limited availability of Zn^{2+} as well as slow diffusion of mIm (shell) side surface 105 was soaked in about 70 mL of neat through defects in the dense layer. FIG. (shell) side surface 105 was soaked in about 70 mL of neat through defects in the dense layer. FIG. 27B reveals that the DI (and replaced three times) to remove the excess Zn^{2} , dual-layer morphology is replaced with a DI (and replaced three times) to remove the excess Zn^{2} dual-layer morphology is replaced with a single dense
Next, about 10 mL heptanes were pushed through the bore 45 membrane layer during the static membrane growth Next, about 10 mL heptanes were pushed through the bore 45 membrane layer during the static membrane growth step.
110 while the outer (shell) side surface 105 was soaked in The discontinuous membrane layer is gradually den procedure was performed with a more volatile hexane **29**C. The information in Table 7 shows a decrease in solution. Then, about 10 mL hexane was pushed through the permeance for both C_2H_2 and C_2H_6 (about 26 G.P.U. solution. Then, about 10 mL hexane was pushed through the permeance for both C_3H_6 and C_3H_8 (about 26 G.P.U. and bore 110 while outer (shell) side surface 105 was soaked in 50 about 1.8 G.P.U., respectively) and an about 70 mL hexane to remove the heptanes. Finally, about
 (C_3H_6) selectivity (about 14.3), indicating that nanoscopic

20 mL of MeOH was pushed through the bore 110 while the

outer (shell) side surface 105 was soaked outer (shell) side surface 105 was soaked in about 70 mL
methanol to remove the DI. The membrane was dried at
about room temperature (e.g., about 25° C.) for about 2 days. 55
Hier the 20 minutes of the second continuous me

measurements were conducted in situ for each arrest point membrane growth step (arrest point #3), a single, continuous $#1$ $#2$ and $#3$ and finally after completion of the IMMP layer ZIF-8 membrane on the inner bore sur #1, #2 and #3, and, finally, after completion of the IMMP layer ZIF-8 membrane on the inner bore surface 110 of the process, the ZIF-8 membrane on the polyamide-imide (e.g., hollow fiber 100 was observed with an overall t process, the ZIF-8 membrane on the polyamide-imide (e.g., hollow fiber 100 was observed with an overall thickness of TORLON®) hollow fiber 100 was removed from the IMMP 60 about 12 μ m (see Table 6). FIGS. 27C & 29D rev TORLON®) hollow fiber 100 was removed from the IMMP 60 reactor module 400 and cross-sectioned for SEM imaging

membrane morphology at the three different arrest points #1, in Table 7 shows a further decrease in permeance for both #2 and #3 shown in FIG. 26B. Tables 6 & 7 show informa- 65 C_3H_6 and C_3H_8 (about 23 G.P.U. and tion about average thickness and C_3H_6/C_3H_8 permeation respectively) (i.e., 1 G.P.U.=3.348×10⁻¹⁰ molm⁻² s⁻¹ Pa⁻¹) characteristics for each of these arrest points. and a further increase in C_3H_6 selectivity

28
After the 120 minutes of the continuous membrane may be observed on the inner bore surface 110 of the hollow
fiber 100 (see FIGS. 27A $\&$ 28) with an overall thickness of Starting with a polyamide-imide (e.g., TORLON®) hol-
w fiber mounted in the IMMP reactor module 400^{12} a ⁵ about 7 um (see Table 6), of which about 4.5 um is a dense

believe this morphology arises from the rapid penetration of rapidly consumed by reaction with Zn^{2+} to form a dense while stirring the solution at about 60 rpm for about 120 $ZIF-8$ membrane layer. However, the formation of this initial minutes to provide a continuous membrane growth step. $_{15}$ dense layer may act as a barrier that re inutes to provide a continuous membrane growth step. 15 dense layer may act as a barrier that restricts the availabilty
The temperature of the reactor module 400 was main-
of mIm species further into the octanol phase, lea tained at about $31.0\pm0.5^{\circ}$ C. throughout the IMMP process. diffusion-limited formation of a second discontinuous film
morphology on top of the dense layer (see FIGS. 29A-29B).

See FIG. 2061. To asset at the densing to the density on top of the dense layer (see FIGS. 29A-29B).

For each arrest point #1, #2 and #3, about 10 mL of neat

To calcular different is effect is different to observe direc

brane growth step .

brane growth step .

After 210 minutes of the static membrane growth step

After the 20 minutes of second continuous membrane α (arrest point #2) during which there was no continuous After the 20 minutes of second continuous membrane $35 \frac{\text{(arrest point \#2) during which there was no continuous}}{\text{replenishment of the limiting reactant }\mathbb{Z}n^{2+},\text{ a single layer}}$ minutes to provide a second static membrane growth step.
A fter the 210 minutes of second static membrane growth step . Figure (see Table 6). The overall ZIF-8 membrane growth rate After the 210 minutes of second static membrane growth μ m (see Table 6). The overall ZIF-8 membrane growth rate
en (arrest point #3), the reaction was stopped. step (arrest point #3), the reaction was stopped.
To stop the reaction about 10 mL of neat 1-octanol during the initial continuous membrane growth step due to To stop the reaction, about 10 mL of neat 1-octanol during the initial continuous membrane growth step due to ly ent was pushed through the bore 110 while the outer limited availability of Zn^{2+} as well as slow dif

Binary propylene $(C_3H_6)/$ propane (C_3H_8) permeation growth step followed by 210 minutes of the second static easurements were conducted in situ for each arrest point reactor module 400 and cross-sectioned for SEM imaging single membrane layer is further densified to a continuous and thickness measurements, as discussed below. membrane layer during the second static membrane growth d thickness measurements, as discussed below. membrane layer during the second static membrane growth FIGS. 27A-27C show the development of the ZIF-8 and second static membrane growth steps. The information and second static membrane growth steps. The information in Table 7 shows a further decrease in permeance for both and a further increase in C_3H_6 selectivity (about 17).

TABLE 7

surements are reliably measuring the intrinsic properties of larger pore size may allow faster formation of a thin ZIF-8
the ZIF-8 membranes with negligible effects of polyamide- membrane layer and better anchoring of this the ZIF-8 membranes with negligible effects of polyamide-
imide $(e, \alpha, TORI ONR)$ bollow fiber support in determing layer to the fiber surface. imide (e.g., TORLON®) hollow fiber support in determin-
ing the defect density and selectivity of the membrane. The 40 Negligible Effects of External Mass Transfer Resistances
fobricated noturinida imida (a.g. TOPI ON®) ha fabricated polyamide-imide (e.g., TORLON®) hollow fibers Further, the inventors confirmed that all permeation mea-
have an outer diameter (OD) of about 300 um and an inner surements are reliably measuring the intrinsic pro have an outer diameter (OD) of about 300 µm and an inner surements are reliably measuring the intrinsic properties of diameter (ID) of about 200 µm and diameter (ID) of about 200 µm and diameter and the ZIF-8 membranes wit

support surfaces, the hollow fibers were immersed in hexane $\frac{45 \text{ (R}_{Total}) \text{ though the membrane was moderate as a result of the same}}{\text{three series resistances: membrane layer (R}_{membrane})\text{, sup-}$ followed by immersion of the saturated fibers in liquid
introgen. Then, the hollow fibers were gently broken into
introgen. Then, the hollow fibers were gently broken into
several pieces, which were then transferred to a measured as 63 ± 5 nm, which is much smaller than the ZIF-8 For an accurate estimate from a bare polyamide-imide
membrane thickness and is not expected to lead to large ϵ . (e.g., TORLON®) hollow fiber that has been ex membrane thickness and is not expected to lead to large $($ e.g., TORLON®) hollow fiber that has been exposed to heterogeneities in the film thickness (see Table 6).

low fibers from SEM images using the Khare and Burris aquesous mlm solution and no Zn^{+2} metal source in the method.²¹ For this purpose (see FIGS. 30A-30B), binary 1-octanol solution (to prevent ZIF-8 formation). images were obtained from cross-sectional SEM images of $\frac{60}{12}$ Using the measured bare hollow fiber permeances along the support via the image analysis program Image 1.2 The with the permeances from the ZIF-8 membr the support via the image analysis program Image 1.2 The with the permeances from the ZIF-8 membrane (Case 1), porosity (c) and average pore size of the fibers were esti-
Equations 1 & 2 were used to back-calculate the

ability of 2.7×10^6 Barrer (i.e., 1 Barrer=1 G.P.U. μ m) at 25° 0.01-5% in the perm
C.

Role of Polyamide-Imide (e.g., TORLON®) Hollow Fiber 35 Without wishing to be bound to any theory, the inventors Further, the inventors confirmed that all permeation mea-
believe that hollow fibers with larger porosity and Further, the inventors confirmed that all permeation mea-

Furtherments are reliably measuring the intrinsic properties of larger pore size may allow faster formation of a thin ZIF-8

diameter (ID) of about 200 µm, as discussed below.
For accuration resistance and below the mass transfer resistances. The total permeation resistance For scanning probe microscopy (SPM) of the hollow fiber $\frac{1}{45}$ ($\frac{1}{R_{Total}}$) through the membrane was modeled as a result of C .

terogeneities in the film thickness (see Table 6).
The inventors also characterized the porosity of the hol-
IMMP approach on the bare hollow fiber 100 with only the The inventors also characterized the porosity of the hol - IMMP approach on the bare hollow fiber 100 with only the low fibers from SEM images using the Khare and Burris agrees and my solution and no Zn^{+2} metal source

mated at 44% and 290 nm, respectively.
The hollow fibers are also highly permeable, as charachines of R_{nembrane} and P_{membrane} (G.P.U.) as shown in Table 8.
The hollow fibers are also highly permeable, as charachines o

$$
R_{membrane} = \frac{1}{P_{membrane}}
$$
 (1)

Single gas permeance results before and after considering the effect of polyamide-imide (e.g., TORLON ®) hollow fiber and external mass transfer resistances at 25° C.					diameter $(1D)$ of about 2/0 μ m and fabricated by a modified spinning. polymer dope compositions and s	
Gas	Measured Permeance (G.P.U.)	Corrected Permeance (G.P.U.)	Error	15	shown in Table 10. TABLE 10	
Propane (C_3H_8) Propylene (C_2H_6)	1.0 66.5	1.0 66.8	0.01% 0.5%		Modified spinning conditions of po TORLON ®) hollow fiber po	

Fabrication of Polyamide-Imide (e.g., TORLON®) Hollow ²⁰ - Fiber Supports

Polyamide imide (e.g., TORLON®) hollow fibers of outer diameter (OD) of about 300 μ m and inner diameter (ID) of about 200 μ m were produced by a spinning method. The polymer dope compositions and spinning conditions are ²⁵ shown in Table 9.

Spinning conditions of polyamide imide (e.g., TORLON ®) hollow fiber porous support.					
Dope composition (PAI/LiNO ₃ /NMP/ ethanol) (wt $\%$)	15/15/66/4				
Dope flow rate (ml/hr)	360	35			
Bore fluid $(NMP/H2O)$ (wt %)	80/20				
Bore fluid flow rate (ml/hr)	120				
Air gap (cm)	16				
Take up rate (m/min)	45				
Operating temperature $(^\circ$ C.)	25				
Quench bath temperature $(^\circ$ C.)	25	40			

to the spinneret compartment. The extruded polymer dope increase the porosity of the hollow fibers, a bore fluid used
passed through an air-gap and into a water quench bath in this experiment comprised NMP and deionized wa passed through an air-gap and into a water quench bath in this experiment comprised NMP and deionized v
(primary coagulant) where the fiber phases separate. A 45 with the weight ratio of 85:15 (instead of 80:20). standard bore fluid used in this experiment comprised a
non-volatile solvent (NMP) and dejonized water (DI) with collected on a rotating take-up drum partially submerged in

Then, the fiber was passed over a TEFLON® guide and
collected on a rotating take-up drum partially submerged in 50 Next, fibers spun under identical conditions were
a second water bath, which was continuously replenished r a second water bath, which was continuously replenished removed from the take-up drum, tied loosely and soaked in with fresh water.

with fresh water.

Next, fibers spun under identical conditions were

removed from the take-up drum, tied loosely and soaked in

DI for about 5 days at room temperature (i.e., about 25° C.)

DI for about 4 days at room te

solvent by immersion for about 1 hour each in three batches $\frac{1 \text{ hour}}{60 \text{ excess method}}$ of fresh methanol (to remove excess water) and then about $\frac{60 \text{ excess method}}{100 \text{ years}}$ 1 hour each in three batches of fresh hexane (to remove After about 2 hours air drying step, the hollow fibers were excess methanol).

After about 2 hours air drying step, the hollow fibers were
dried at about 130° C. for about 24 hours to completely
Fig. 2266 nm which is similar to the previous hollow fibers
had an RMS roughness of
remove any residual N

and is not expected to lead to large heterogeneities in the film thickness (see Table 6). Further, the porosity and average pore size of the hollow fibers were estimated at 44% and 290 nm, respectively (see FIGS. 30A-30B).

 $R_{Total} = R_{membrane} + R_{Torton + external$ (2) am, respectively (see FIGS. 30A-30B).

5 Fabrication of Improved Polyamide Imide (e.g., TOR-

LON®) Hollow Fiber Supports

The improved polyamide imide (e.g., $TORLON®$) hollow TABLE 8 fibers of outer diamer (OD) of about 360 µm and inner
diameter (ID) of about 270 µm and enhanced porosity were 10 fabricated by a modified spinning approach. The modified polymer dope compositions and spinning conditions are shown in Table 10.

0.01% Propane (C_3H_8) 1.0 $1.0\,$ Propylene (C_3H_6) 66.5 66.8 0.5%	Modified spinning conditions of polyamide imide (e.g., TORLON [®]) hollow fiber porous support.			
Hydrogen (H_2) 1596 3.7% 1537 cation of Polyamide-Imide (e.g., TORLON®) Hollow - 20	Dope composition $(PALLINO3/NMP/$ ethanol $)$ (wt %)	15/15/66/4		
Supports yamide imide (e.g., TORLON®) hollow fibers of outer ter (OD) of about 300 µm and inner diameter (ID) of 200 um were produced by a spinning method. The ner dope compositions and spinning conditions are 25	Dope flow rate (ml/hr) Bore fluid $(NMP/H2O)$ (wt %) Bore fluid flow rate (ml/hr) Air gap (cm) Take up rate (m/min) Operating temperature $(^\circ$ C.) Quench bath temperature $(^\circ$ C.)	300 85/15 100 15 40 27 25		

TABLE 9 Non-volatile solvent (NMP), non-solvent (water) and pore formers ($LiNO₃$) in the dope composition, quench bath temperatures (i.e., about 25° C.), take-up rate (i.e., about 40 m/minute) and air gap height (i.e., about 15 cm) were applied in order to identify a robust process to produce improved hollow fibers with engineered surface porosity. The key parameters control the morphology of the improved hollow fibers include composition of the polymer dope, composition of the bore liquid, height of the air gap and spinning speed.

A polyamide-imide (e.g., TORLON®) core dope was fed
to the spinneret compartment. The extruded polymer dope passed through an air-gap and into a water quench bath A polyamide-imide (e.g., TORLON®) core dope was fed (primary coagulant) where the fiber phases separate. To the spinneret compartment. The extruded polymer dope increase the porosity of the hollow fibers, a bore fluid used

non-volatile solvent (NMP) and deionized water (DI) with collected on a rotating take-up drum partially submerged in
a second water bath, which was continuously replenished

promote complete removal of the water-soluble $LINO_3$. Solvent by immersion for about 1 hour each in three batches Then, the water in the hollow fiber was exchanged with a of fresh methanol (to remove excess water) and then about $\frac{1}{2}$ hour each in three batches of fresh hexane (to remove

move any residual NMP.
The hollow fibers had an RMS roughness of 63 ± 5 nm, (discussed above) and, thus, provides a flat surface for ZIF-8 The hollow fibers had an RMS roughness of 63 ± 5 nm, (discussed above) and, thus, provides a flat surface for ZIF-8 which is much smaller than the ZIF-8 membrane thickness membrane layer formation. However, the porosity membrane layer formation. However, the porosity and averlarger at 55% and 480 nm, respectively (see FIGS. 30C-
30D). Cases 1 2 and 3 denicting uniform membrane layers for

uses a fabricated polyamide-imide (e.g., TORLON®) hol-
low ther and an isothermal IMMP annmach: Case 2 uses an lower than for Case 1 and Case 2 (about 10 µm, and about low fiber and an isothermal IMMP approach; Case 2 uses an lower than for Case 1 and Case 2 (about 10 µm, and about improved polyamide-imide (e.g., TORLON®) hollow fiber 10 9 µm, respectively) indicating that the initial h improved polyamide-imide (e.g., TORLON®) hollow fiber 10^{-9} µm, respectively) indicating that the initial heating IMMP prepared using a modified spinning technique and a modi-
approach remains effective in reducing the prepared using a modified spinning technique and a modi-
fied initial heating IMMP approach; and Case 3 uses the mess on the improved hollow fiber support.

400,¹² about 10 mL of neat 1-octanol was flowed at about 2 with the reactor module 400 directly acting as a permeation mL/min through a bore 110 of a hollow fiber 100 using a $_{20}$ cell. See FIG. 16. The separation pro syringe pump. Then, about 3 mL of a Zn+2/1-octanol membranes were also characterized by $H_2/C_3H_6/C_3H_8$ ter-
solution containing about 0.22 σ Zinc nitrate hexahydrate tiary equimolar mixture permeation as a function solution containing about 0.22 g Zinc nitrate hexahydrate than equimolar mixture permeation as a function of tem-
(Zn⁺²) in about 40 mL 1-octanol (about 0.018 mol/L) was perature. In an embodiment, the feed mixture may (Zn^{+2}) in about 40 mL 1-octanol (about 0.018 mol/L) was perature. In an embodiment, the feed mixture may be flowed at about 0.6 mL/hour through the bore 110 of the selected from the group consisting of hydrogen/hydrocar flowed at about 0.6 mL/hour through the bore 110 of the selected from the group consisting of hydrogen/hydrocar-
hollow fiber 100 while an aqueous mum solution containing 25 bons, hydrogen/propane, propylene/propane and bu hollow fiber 100 while an aqueous mum solution containing 25 about 9 g of mum in about 80 mL deionized water (DI)
(about 1.37 mol/L) was poured into the reactor chamber 345
immersing the outer (shell) side surface 105 of the hollow
fiber 100 while stirring the solution at about 60

The temperature of reactor module 400 was set and maintained at about 30° C.

the flow rate of about 0.6 mL/hour through the bore 110 of (shell) side surface 105 of the hollow fiber 100 at about 30
the hollow fiber 100 was stopped for about 210 minutes to $\frac{mL}{m}$ minute. The injected gases were the hollow fiber 100 was stopped for about 210 minutes to provide a static membrane growth step.

flow rate of about 0.6 mL/hour through the bore 110 of the 40 higher feed pressures (up to 6 bar), exchollow fiber 100 was continued for about 20 minutes to for high-pressure operation were used.

provide a second continuous membrane growth step.

An online gas chromatograph (Shimadzu GC-2014) with

step, the flow rate of about 0.6 mL/hour through the bore 110

of the hollow fiber 100 was stopped for about 210 minut to provide a second static membrane growth step.
After the 210 minute second static membrane growth

while the outer (shell) side surface 105 was soaked in about fitting. The pressure differential between the bore side 110
70 mL of neat DI (and replaced three times) to remove the and the shell side 105 was, thus, controll 70 mL of neat DI (and replaced three times) to remove the and the shell side 105 was, thus, controlled from 0 psi to 90 excess Zn^{+2} . Next, about 10 mL heptanes were flowed at psi helium while the shell side 105 he excess Zn^{+2} . Next, about 10 mL heptanes were flowed at psi helium while the shell side 105 helium flow rate was about 2 mL/min through the bore 110 while the outer (shell) measured using a digital flow meter. Then, th about 2 mL/min through the bore 110 while the outer (shell) measured using a digital flow meter. Then, the bore 110 of side surface 105 was soaked in about 70 mL heptanes to 55 the ZIF-8 membrane was saturated with Fluori side surface 105 was soaked in about 70 mL heptanes to 55 the ZIF-8 membrane was saturated with Fluorinert (FC-40) remove the 1-octanol. Next, similar procedure was per-
colution that has a low surface tonsion. Then the remove the 1-octanol. Next, similar procedure was per-
formed with a more volatile hexane solution. Then, about 10
mL hexane was flowed at about 2 mL/min through the bore
110 while outer (shell) side surface 105 was soaked 110 while outer (shell) side surface 105 was soaked in about Atomic force microscopy (AFM) images of inner surfaces
20 mJ hovens to remove the hertanes. Finally, shout 20 mJ so of the hollow fibers were obtained with an IC 70 mL hexane to remove the heptanes. Finally, about 20 mL 60 ^{of the} hollow fibers were obtained with an ICON Dimen-
of MeOH was flowed at about 2 mL/min through the hore sion® scanning probe microscope (Bruker). The AFM of MeOH was flowed at about 2 mL/min through the bore sion® scanning probe microscope (Bruker). The AFM was so
110 while the outer (shell) side surface 105 was soaked in operated under tapping mode with Mikromasch NSC14 110 while the outer (shell) side surface 105 was soaked in operated under tapping mode with Mikromasch NSC14
about 70 mL methanol to remove the DL The membrane was silicon cantilevers (8 nm tip radius, 5 N/m force constant about 70 mL methanol to remove the DI. The membrane was
direct cantrievers (8 nm up radius, 5 N/m force constant,
dried at about room temperature (e.g., about 25° C.) for
about 1 do kHz typical resonance frequency).
dout

membranes grown under various IMMP process conditions

age pore size of the improved hollow fibers were much are shown in FIGS. 31A-32C. FIGS. 31A-31C shows cross-
larger at 55% and 480 nm, respectively (see FIGS. 30C-
sectional SEM images of the ZIF-8 membranes obtained in 30D). Cases 1, 2 and 3, depicting uniform membrane layers for
To investigate the effect of modified hollow fibers in
conjunction with a modified IMMP approach, the inventors
prepared ZIF-8 membrane thickness at different

fied initial heating IMMP approach; and Case 3 uses the

improved polyamide-imide (e.g., TORLON®) hollow fiber

and the modified initial heating IMMP approach.

Performance in Membrane Synthesis Using IMMP Reactor

Perfor Starting with a fabricated polyamide-imide (e.g., TOR-
LON®) hollow fiber mounted in the IMMP reactor module
 $\frac{100^{12} \text{ about 10 mJ}}{400^{12} \text{ about 10 mJ}}$ of next 1 octavel was flowed at about 2 with the reactor module 400 dir

growth step. The fabrication of the hollow fiber 100 is
discussed below.
The temperature of reeder module 400 wes set and LON®) hollow fiber 100 at a precisely controlled flow rate (e.g., about 10 mL/minutes) via a mass flow controller (MFC) while an argon sweep gas was flowed across an outer After the 120 minute continuous membrane growth step, $35 \frac{(MFC)}{N}$ while an argon sweep gas was flowed across an outer $16 \frac{MFC}{N}$ (shell) side surface 105 of the hollow fiber 100 at about 30 before entering the feed side of the reactor/permeation module. A similar apparatus was used for measurements at After the 210 minute static membrane growth step, the module. A similar apparatus was used for measurements at we rate of about 0.6 mL/hour through the bore 110 of the 40 higher feed pressures (up to 6 bar), except that MF

After the 210 minute second static membrane growth permporosimetry equipment with the IMMP reactor module step, the reaction was stopped. 400 directly acting as a permeation cell. First, helium was 400 directly acting as a permeation cell. First, helium was To stop the reaction, about 10 mL of neat 1-octanol introduced into one end of the bore 110 of the ZIF-8 solvent was flowed at about 2 mL/min through the bore 110 so membrane while plugging the other end with a Swagelok

Without wishing to be bound by any theory, the inventors about 2 days.
lieve that increasing the rate of formation of the initial The initial heating IMMP approach increased the rate of believe that increasing the rate of formation of the initial The initial heating IMMP approach increased the rate of dense harrier layer may significantly reduce its thickness 15 formation of the initial dense layer. See F dense barrier layer may significantly reduce its thickness 15 formation of the initial dense layer. See FIGS 27D-27F. A
considerably thinner (about 2.5 µm) dense membrane layer during the initial continuous membrane growth step, since a considerably thinner (about 2.5 µm) dense membrane layer
is formed initially, leading to thinner final membrane thick-
is formed initially, leading to thinner fin more rapidly formed barrier layer would effectively inhibit is formed initially, leading to thinner final membrane thick-
further reaction limited dense film growth. This would result ness (about 8 µm) with a corresponding further reaction-limited dense film growth. This would result
in a reduced overall membrane thickness and, hence, a permeance (about 33 G.P.U.) and no loss of selectivity over

LON®) hollow fiber mounted in the IMMP reactor module 25 (e.g., TORLON®) hollow fiber mounted in the IMMP 400,¹² about 10 mL of neat 1-octanol was flowed at about 2 reactor module 400.¹² about 10 mL of neat 1-octanol 400,¹² about 10 mL of neat 1-octanol was flowed at about 2 reactor module 400 ,¹² about 10 mL of neat 1-octanol was mL/min through a bore 110 of a hollow mL/min through a bore 110 of a hollow fiber 100 using a flowed at about 2 mL/min through a bore 110 of a hollow
syringe pump. Then, about 3 mL of a Zn+2/1-octanol fiber 100 using a syringe pump. Then, about 3 mL of a syringe pump. Then, about 3 mL of a Zn+2/1-octanol fiber 100 using a syringe pump. Then, about 3 mL of a solution containing about 0.22 g Zinc nitrate hexahydrate $Zn+2/1$ -octanol solution containing about 0.22 g Zinc nit (Zn^{+2}) in about 40 mL 1-octanol (about 0.018 mol/L) was 30 hexahydrate (Zn^{+2}) in about 40 mL 1-octanol (about 0.018 flowed at about 0.6 mL/hour through the bore flowed at about 0.6 mL/hour through the bore 110 of the hollow fiber 100 while an aqueous mum solution containing hollow fiber 100 while an aqueous mum solution containing 110 of the hollow fiber 100 while an aqueous mum solution about 9 g of mum in about 80 mL deionized water (DI) containing about 9 g of mum in about 80 mL deionized about 9 g of mum in about 80 mL deionized water (DI) containing about 9 g of mum in about 80 mL deionized water (about 1.37 mol/L) was poured into the reactor chamber immersing the outer (shell) side surface 105 of the hollow 35 345 immersing the outer (shell) side surface 105 of the fiber 100 while stirring the solution at about 60 rpm for hollow fiber 100 while stirring the solution a fiber 100 while stirring the solution at about 60 rpm for about 120 minutes to provide a continuous membrane for about 120 minutes to provide a continuous membrane growth step. The fabrication of the improved hollow fiber growth step. The fabrication of the hollow fiber 100 is growth step. The fabrication of the improved hollow fiber discussed above.
100 is discussed above.

The temperature of the reactor module 400 was initially 40 The temperature of the reactor module 400 was initially set at about 22° C. During the continuous membrane growth set at about 22° C. During the continu step, the temperature of the reactor module 400 was increased linearly from about 22° C. to about 42° C. in about increased linearly from about 22° C. to about 42° C. in about increased linearly from about 22° C. to about 42° C. in about 25 minutes (about 0.8 ° C./min) to increase the formation of 25 minutes (i.e., at about 0.8 ° C 25 minutes (about 0.8° C./min) to increase the formation of 25 minutes (i.e., at about 0.8° C./min) Then, the temperature the initial dense membrane layer. Then, the temperature of 45 of the reactor module 400 was decreas the reactor module 400 was decreased from about 42° C. to to about 30° C. in about 60 minutes (i.e., about 0.2° C./min) about 30° C. in about 30° C. in about 30° C. in about 30° C. in ab

the flow rate of about 0.6 mL/hour through the bore 110 of the hollow fiber 100 was stopped for about 210 minutes to the hollow fiber 100 was stopped for about 210 minutes to provide a static membrane growth step.

flow rate of about 0.6 mL/hour through the bore 110 of the 55 hollow fiber 100 was continued for about 20 minutes to provide a second continuous membrane growth step.

After the 20 minute second continuous membrane growth step, the flow rate of about 0.6 mL/hour through the bore 110 step, the flow rate of about 0.6 mL/hour through the bore 110 of the hollow fiber 100 was stopped for step, the flow rate of about 0.6 mL/hour through the bore 110 of the hollow fiber 100 was stopped for about 210 minutes of the hollow fiber 100 was stopped for about 210 minutes 60 to provide a second static membrane growt

to provide a second static membrane growth step.
After the 210 minute second static membrane growth
After the 210 minute second static membrane growth
step, the reaction was stopped.
To stop the reaction was stopped.
To st

solvent was flowed at about 2 mL/min through the bore 110 while the outer (shell) side surface 105 was soaked in about while the outer (shell) side surface 105 was soaked in about 70 mL of neat DI (and replaced three times) to remove the 70 mL of neat DI (and replaced three times) to remove the excess Zn^{+2} . Next, about 10 mL heptanes

TABLE 11 excess Zn^{+2} . Next, about 10 mL heptanes were flowed at about 2 mL/min through the bore 110 while the outer (shell) side surface 105 was soaked in about 70 mL heptanes to remove the 1-octanol. Next, a similar procedure was per-
5 formed with a more volatile hexane solution. Then, about 10 mL hexane was flowed at about 2 mL/min through the bore 110 while outer (shell) side surface 105 was soaked in about 70 mL hexane to remove the heptanes. Finally, about 20 mL of MeOH was flowed at about 2 mL/min through the bore 10 110 while the outer (shell) side surface 105 was soaked in about 70 mL methanol to remove the DI. The membrane was dried at about room temperature (e.g., about 25° C.) for

in a reduced overall membrane thickness and, hence, a
higher permeance.
Performance in Membrane Synthesis Using IMMP Reactor
Performance in Membrane Synthesis Using IMMP Reactor
Cell/Module: Initial Heating
Cell/Module: Im

 $Zn+2/1$ -octanol solution containing about 0.22 g Zinc nitrate (DI) (about 1.37 mol/ L) was poured into the reactor chamber 345 immersing the outer (shell) side surface 105 of the

set at about 22° C. During the continuous membrane growth step, the temperature of the reactor module 400 was

module was set and maintained at about 30° C. After the 120 minute continuous membrane growth step,
After the 120 minute continuous membrane growth step, 50 the flow rate of about 0.6 mL/hour through the bore 110 of
the fl

provide a static membrane growth step.

After the 210 minute static membrane growth step . provide a static membrane growth step . After the 210 minute static membrane growth step, the . After the 210 minute static membran flow rate of about 0.6 mL/hour through the bore 110 of the hollow fiber 100 was continued for about 20 minutes to

provide a second continuous membrane growth step.
After the 20 minutes to provide a second continuous membrane growth step, the flow rate of about 0.6 mL/hour through the bore 110

solvent was flowed at about 2 mL/min through the bore 110 while the outer (shell) side surface 105 was soaked in about excess Zn^{+2} . Next, about 10 mL heptanes were flowed at

 37 about 2 mL/min through the bore 110 while the outer (shell) side surface 105 was soaked in about 70 mL heptanes to imide (e.g., TORLON®) hollow fibers with separation fac-
remove the 1-octanol. Next, a similar procedure was per-
tors of about 370 at 120° C. and about 12 at 2 formed with a more volatile hexane solution. Then, about 10 respectively).²⁵ This data shows a strong temperature depen-
mL hexane was flowed at about 2 mL/min through the bore 5 dence of the C_oH_c permeance, resulti mL hexane was flowed at about 2 mL/min through the bore 5 dence of the C_3H_6 permeance, resulting in a moderate 110 while outer (shell) side surface 105 was soaked in about decrease in separation factor as temperature 110 while outer (shell) side surface 105 was soaked in about

70 mL hexane to remove the heptanes. Finally, about 20 mL

of MeOH was flowed at about 2 mL/min through the bore

110 while the outer (shell) side surface 105

Fo compare the performance of these ZH⁻¹ membranes,
equimolar C₃H₆/C₃H₈ mixed-gas permeation data were col-
lected using a steady-state Wicke-Kallenbach technique at 15 to effectively separate C₃H₆ from C₃ 25° C. with the IMMP reactor directly acting as a permeation with separation factor of about 53 at 25° C. It is also noted module (see Table 11). The Case 3 ZIF-8 membrane shows that these Case 3 membranes can be utilized much higher C_3H_6 permeance than the Cases 1 and 2 from C_3H_6 (i.e., H_2/C_3H_6 separation factor of 99 \pm 3 at 120° membranes due to a much lower thickness. Also, the Case 3 C., which industrially attractive).¹⁹ membranes due to a much lower thickness. Also, the Case 3 C., which industrially attractive).¹⁹
membrane also shows much higher C_3H_8/C_3H_8 selectivity 20 Further, the Case 3 ZIF-8 membrane was tested for binary

believe that the ZIF-8 membrane defects are decreased due minor (about 13-15%) reduction at 6 bar. The C_3H_6 perto better anchoring of the membrane layer on the more meance shows a reduction of about 25%, which is expe porous hollow fiber support. The microstructural changes 25 in molecular sieve materials due to pore saturation effects.
were investigated in more detail with helium permporosim-
However, it is important to note that the etry.²² The helium flux was measured under dry conditions (throughput) at 6 bar is about 3.75 to 4 times higher due to and, then, after wetting with Fluorinert (FC-40) liquid that the much greater driving force. has a low surface tension, helium permeance is measured as In conclusion, the mechanistic findings and resulting a function of pressure in the feed gas. Under wetted condi- 30 significant improvements in membrane performan tions, ZIF-8 membranes that have mesopore defects such as ports a two-stage separation of ternary H_2/C_3H_8 mix-
pinholes or cracks would show a sudden increment in tures (e.g., a first ZIF-8 membrane stage of small tot pinholes or cracks would show a sudden increment in tures (e.g., a first ZIF-8 membrane stage of small total area
helium permeance at a pressure that is sufficient to displace that takes advantage of the very high H₂ per helium permeance at a pressure that is sufficient to displace that takes advantage of the very high H_2 permeance to the wetting liquid from the mesopore defects. The defect separate it from C_3H_6 and C_3H_8 , follo size can then be estimated by the Cantor equation.²³ How- 35 ever, such phenomena are not observed in the pressure range ever, such phenomena are not observed in the pressure range C_3H_8). Such two-stage separation is possible due to the 0-90 psi for any of the Case 1-3 ZIF-8 membranes (see FIG. significant advancements of the present in 0-90 psi for any of the Case 1-3 ZIF-8 membranes (see FIG. significant advancements of the present invention, first, the 33), showing that continuous membrane layers are formed modification of the IMMP approach to include 33), showing that continuous membrane layers are formed modification of the IMMP approach to include an initial over the entire fiber with no large defects (i.e., defects less heating growth step and, secondly, the enginee

than 20 nm in diameter, length or width at 90 psi helium). 40 modified hollow fiber support with enhanced surface prop-
However, the dry helium permeabilities (see FIG. 33) also erties.
allow the qualitative characterizati (i.e., less than 20 nm). The ideal range of helium perme-
ability from a ZIF-8 membrane free of defects is represented 95 mol % hydrogen, about 2 mol % to about 95 mol % by the shaded area in FIG. 33, which is obtained using the 45 range of corrected diffusivity and adsorption isotherm parameters for helium in ZIF-8 as given by Zhang et al.²⁴ The Case 1 ZIF-8 membranes display significantly higher The Case 1 ZIF-8 membranes display significantly higher % to about 5 mol % n-propane, about 0 mol % to about 5 mol % of about 5 mol dry helium permeability outside the ideal region, clearly % i-propane, about 0 mol % to ab indicating the presence of nanoscopic defects. However, the 50 2 mol % to about 95 mol % i-butane, about 2 mol % to abou dry helium permeability progressively declines and reaches the ideal region in Case 2 and Case 3 ZIF-8 membranes, the ideal region in Case 2 and Case 3 ZIF-8 membranes, pentenes and about 0 mol % to about 15 mol % pentanes and indicating that both membrane thickness and defect density mixtures thereof. have been reduced. The Case 3 ZIF-8 membrane has the In a hydrocarbon separation embodiment, an operating lowest dry helium permeability, corresponding to the most 55 temperature for the reactor cell 400 is from about 30° defect-free membranes which also has the best separation about 120° C. or any value there between. In an embodi-
performance as shown in Table 11.

mixture separation properties of the Case 3 ZIF-8 mem-
branes in more detail as a function of temperature from ω In a hydrocarbon separation embodiment, an operating
about 25° C. to about 120° C. FIG. 34A shows equimol about 25° C. to about 120° C. FIG. 34A shows equimolar pressure for the reactor cell 400 is from about 1 bar to about binary (H_2/C_3H_8) and $C_3H_6/C_3H_8)$ mixture permeation prop- 14 bar or any value there between. In binary $(H_2/C_3H_8$ and $C_3H_6/C_3H_8)$ mixture permeation prop-
eration or any value there between. In an embodiment, the
erties. Clear molecular sieving effects are observed for the operating pressure is about 1 bar. In improved ZIF-8 membranes on polyamide-imide (e.g., operating pressure is about 4 bar. In an embodiment, the TORLON®) hollow fibers with H_2/C_2H_8 and C_3H_6/C_3H_8 65 operating pressure is about 7 bar. In an embodiment TORLON®) hollow fibers with H_2/C_3H_8 and C_3H_6/C_3H_8 65 separation factors of about 2000 at 70° C. and about 65 at 25° C., respectively. This performance is much improved

38
from previously reported ZIF-8 membranes on polyamide-

due to a significant drop in the C_3H_8 permeance.
Without wishing to be bound by any theory, the inventors bar at 25° C. The selectivity remains high and shows only

significant improvements in membrane performance supseparate it from C_3H_6 and C_3H_8 , followed by a second ZIF-8 membrane stage of larger area that separates C_3H_6 from heating growth step and, secondly, the engineering of a

95 mol % hydrogen, about 2 mol % to about 95 mol % i-butane, about 2 mol % to about 95 mol % n-butane and mixtures thereof. For example, the feed mixture may comprise about 0 mol % to about 2 mol % ethane, about 0 mol % i-propane, about 0 mol % to about 5 mol % butenes, about

performance as shown in Table 11. The set of the operating temperature is about 30° C. In an Finally, the inventors characterized the binary and ternary embodiment, the operating temperature is about 70° C. In a embodiment, the operating temperature is about 70° C. In an

> operating pressure is about 1 bar. In an embodiment, the operating pressure is about 4 bar. In an embodiment, the operating pressure is about 10 bar. In an embodiment, the operating pressure is about 14 bar.

15

presented to best explain the present invention and its practical application and to thereby enable those skilled in As used herein, the phrase "nanocrystals" means the seed the art to make and utilize the invention. However, those crystals have an average size of less than one skilled in the art will recognize that the foregoing descrip- $\frac{5}{2}$ preferably about 400-600 nm, and a size distribution of tion and examples have been presented for the purpose of about $\pm 10\%$. For polymers with v tion and examples have been presented for the purpose of about $\pm 10\%$. For polymers with very small pores, a smaller illustration and example only. The description as set forth is seed crystal may be needed, and therefo illustration and example only. The description as set forth is
not intended to be exhaustive or to limit the invention to the
precise form disclosed. Many modifications and variations
are precise form disclosed. Many modif are possible in light of the above teaching without departing $\frac{10}{10}$ in which the MOF precursors have a low solubility, such that from the spirit and scope of the following claims. The $\frac{10}{10}$ on introducing the

have the same open-ended meaning as "comprising," "com-
prises," and "comprise," provided above.
As used herein, the terms "including," "includes," and
"include" have the same open-ended meaning as "compris-
ing," "compris

before the term to one or more material elements recited after the term, where the material element or elements listed after the transition term are the only material elements that make up the subject. 45

As used herein, the term "free of defects" means that the metal-organic framework (MOF) membrane coating is from about 95% to about 100% free of defects (and any range or value there between), and, preferably, from about 97% to about 100% free of defects or about 100% free of defects, 50 and that any existing defect sizes is less than about 20 nanometer (and any range or value there between) in diameter, length or width, and, preferably, the defect sizes are less than 10 nanometers in diameter, length or width such that the membrane coating is essentially continuous to prevent gas or 55 liquid being treated from leaking through a defect (e.g., pore, crack, void, etc.). INCORPORATION BY REFERENCE

As used herein, the phrase "growth solvent" means a liquid in which the MOF seed crystals can be increased in All patents and patent applications, articles, reports, and

As used herein, the term "mesoporous" means having a invention, as follows:

3D structure of interconnected pores, wherein the pore sizes 1) J. Gascon, et al., CHEM. MATER. 24 (2012) 2829-2844. range in diameter from about 0.1 nanometer to about 10 2) K. Varoon, et al., SCENCE 334(6052) (2011) 72-75.
nanometers (and any range or value there between), and, 65 3) M. Shah, et al., IND. ENG. CHEM. RES. 51 (2012) 2179 nanometer to about 5 nanometers or from about 2 nanome - 4) M. G. Buonomenna, RSC ADVANCES 3 (2013) 5694-5740.

The embodiments and examples set forth herein are ters to about 4 nanometers. However, the pores sizes may be esented to best explain the present invention and its varied depending on the gases to be separated.

from the spirit and scope of the following claims. The
invention is specifically intended to be as broad as the claims
below and their equivalents.
 $\frac{1}{100}$ a large number of nuclei, which are the seeds for growing a
la

Definitions $\frac{15}{15}$ and $\frac{15}{15}$ as used herein, the term "polymer" means polymers made from one or more monomeric unit, and, thus, the term

As used herein, the terms "a," "an," "the," and "said"

means one or more more more and state decoted the second means one or more and state decoted in the second means one or more, unless the control of means the control

size, either by crystal deposition or synthesis, but without 60 other documents cited herein are fully incorporated by harming the polymer.
As used herein, the term "mesoporous" means having a invention, as follows:

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- 15) K. Li, et al., J. Am. CHEM. Soc. 131(30) (2009) 10368-15 opposing the third hole from the second surface.
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- 19) Y. Pan, Z. Lai, CHEM. COMMUN. 47(37) (2011) 10275- 20
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- $2130-2134$.

2130-2134.

2130-2130 2130 214 .

2130-214 . The Pressure of also School of the Server 245(6102) (2014) a reactor module having a base shape and a first height,
- 35

1. A reactor cell for flow processing molecular sieving membranes, comprising:

- wherein the base shape is selected from the group consisting of square, rectangular, circular and ellipse; extends into the reactor chamber from a first surface, a 45 second hole opposing the first hole extends into the and a fourth hole opposing the third hole extends into the reactor chamber from a fourth surface; wherein the 50 the reactor chamber from a fourth surface; wherein the 50 a reactor module cover having the same base shape as the first hole is fluidically connected to a first inlet, the reactor module and a second height, wherein the r third hole is fluidically connected to a second inlet and the fourth hole is fluidically connected to a second the fourth hole is fluidically connected to a second a hollow fiber having a first end and a second end, wherein outlet:

s a length of the first end is supported by and sealed into
-
- a hollow fiber having a first end and a second end, wherein 60 brane that is uniform and at least 95% free of defects a length of the first end is supported by and sealed into grown on an inner bore surface of the hollow f the first hole and a length of the second end is supported making the reactor cell device comprising the steps of:
by and sealed into the second hole; wherein the first and fabricating the reactor chamber extending into th by and sealed into the second hole; wherein the first and fabricating the reactor chamber extending into the reactor second ends of the hollow fiber are capped with a module from the upper surface of the reactor module; second ends of the hollow fiber are capped with a module from the upper surface of the reactor module; capping solution, wherein a molecular sieving mem- 65 fabricating an O-ring groove with an inner dimension brane that is uniform and at least 95% free of defects is slightly larger than and of grown on an inner bore surface of the hollow fiber. of the reactor chamber; grown on an inner bore surface of the hollow fiber.

42

5) M. Tsapatsis, SCENCE 334(6057) (2011) 767-768.
 2. The device of claim 1, wherein the molecular sieving

6) T. Cao, et al., SCENCE 334(6062) (2011) 1533-1538.

T) J. Choi, et al., SCENCE 325(5940) (2009) 590-593.

an 7) J. Choi, et al., SCIENCE 325 (5940) (2009) 590-593. any defects are less than about 20 nm in diameter, length or 8) Y. Pan, et al., J. MEMBR. SCI. 421-422 (2012) 292-298. width.

- (8) Y. Pan, et al., J. MEMBR. SCI. 421-422 (2012) 292-298. width.

9) J. A. Thompson, et al., CHEM. MATER. 24 (2012) 1930-5

1936.

10) K. S. Park, et al., PROC. NATL. ACAD. SCI. U.S.A. 103(27)

2006) 10186-10191.

11) A.
- 13) R. Ameloot, et al., NAT. CHEM. 3 (2011) 382-387. from the second surface, the third hole extends into the 14) R. Ameloot, et al., NAT. CHEM. 3 (2016) 401 400 14) M. Pera-Titus, et al., J. MEMBR. Sci. 278 (2006) 401-409.
15) K. Li, et al., J. AM. Curm, Sco. 131(30) (2000) 10368, 15 opposing the third hole extends into the reactor chamber

10369. from the second surface.

16 K. S. Jang, et al., CHEM. MATER. 23 (2011) 3025-3028. 5. The device of claim 1, wherein the first hole extends

17) Y Pan et al. I MEMBR. SCL 390-391 (2012) 93-98. into the reactor chamb 17) Y. Pan, et al., J. MEMBR. Sci. 390-391 (2012) 93-98. into the reactor chamber from the first surface, the second 18) H. Bux et al. CHEM MATER 23 (2011) 2262-2269 hole opposing the first hole extends into the reactor c 18) H. Bux, et al., CHEM. MATER. 23 (2011) 2262-2269. hole opposing the first hole extends into the reactor chamber 19) Y. Pan. Z. Lai. CHEM. COMMUN. $47(37)$ (2011) 10275- 20 from the second surface, the third hole exten 10277.

10277. reactor chamber from an upper surface of a reactor module

10277. EWOO, et al., CHEM. COMMUN. 49 (2013) 3854-3856. cover and the fourth hole opposing the third hole extends 20) H. T. Kwon, et al., CHEM. COMMUN. 49 (2013) 3854-3856. cover and the fourth hole opposing the third hole extends 21) H. S. Khare, et al., POLYMER 51 (2010) 719-729. into the reactor chamber from a bottom surface of t 21) H. S. Khare, et al., POLYMER 51 (2010) 719-729. into the reactor chamber from a bottom surface of the reactor 22) W. S. Rasband, "*ImageJ*", U.S. National Institutes of chamber.

22 **6.** The device of claim 1, wherein the lengths of the first 23) D. Korelskiy, et al., J. MEMBR. SCL. 417-418 (2012) and second ends of the hollow fiber are sealed into the first 23 D. Korelskiy, et al., J. MEMBR. Sci. 417-418 (2012) and second ends of the hollow fiber are sealed into the first and second holes with an adhesive

183-192.

24) J. Kärger, et al., DIFFUSION IN ZEOLITES AND OTHER MICROPO-

ROUS SOLIDS (Wiley, 1992).

25) J. I. Calvo, et al., J. COLLOID & INTERFACE SCI. 176(2) 30 (PDMS) in heptane.

26) C. Zhang, et al., J. PHYS. CHEM

- 27) A. J. Brown, et al., S. Nair, SCIENCE 345(6192) (2014)

72-75.

What is alsimed is:

a reactor module having a base shape and a first neight,
 $\frac{1}{25}$
 $\frac{1}{25}$
 $\frac{1}{25}$

What is alsimed is: What is claimed is:

What is claimed is:
 $\frac{1}{2}$ A reactor cell for flow processing molecular signing

Wherein a first hole

module from an upper surface; wherein a first hole extends into the reactor chamber from a first surface, a second hole opposing the first hole extends into the a reactor module having a base shape and a first height, 40 second hole opposing the first hole extends into the
wherein the base shape is selected from the group reactor chamber from a second surface, a third hole consisting of square, rectangular, circular and ellipse; extends into the reactor chamber from a third surface
wherein a reactor chamber extends into the reactor and a fourth hole opposing the third hole extends into wherein a reactor chamber extends into the reactor and a fourth hole opposing the third hole extends into module from an upper surface; wherein a first hole the reactor chamber from a fourth surface; wherein the the reactor chamber from a fourth surface; wherein the first hole is fluidically connected to a first inlet, the second hole opposing the first hole extends into the second hole is fluidically connected to a first outlet, the reactor chamber from a second surface, a third hole is third hole is fluidically connected to a second inlet reactor chamber from a second surface, a third hole third hole is fluidically connected to a second inlet and extends into the reactor chamber from a third surface the fourth hole is fluidically connected to a second the fourth hole is fluidically connected to a second outlet;
	- first hole is fluidically connected to a first inlet, the reactor module and a second height, wherein the reactor second hole is fluidically connected to a first outlet, the module cover is fastened to the reactor module t module cover is fastened to the reactor module to seal
the reactor chamber; and
	- outlet;
a reactor module cover having the same base shape as the the first hole and a length of the second end is supported
the first hole and a length of the second end is supported reactor module cover having the same base shape as the the first hole and a length of the second end is supported reactor module and a second height, wherein the reactor by and sealed into the second hole; wherein the firs reactor module and a second height, wherein the reactor by and sealed into the second hole; wherein the first and module cover is fastened to the reactor module to seal second ends of the hollow fiber are capped with a module cover is fastened to the reactor module to seal second ends of the hollow fiber are capped with a the reactor chamber; and capping solution, wherein a molecular sieving memcapping solution, wherein a molecular sieving membrane that is uniform and at least 95% free of defects is
		-
		- fabricating an O-ring groove with an inner dimension slightly larger than and offset from the outer dimension

30

- fabricating the first hole extending into the reactor cham-
the method of claim 8, further comprising the steps
the first surface and the second hole opposing
the first hole and extending into the reactor chamber
a. flowin
- fabricating a third hole extending into the reactor chamber 5 second temperature during a portion of the first period;
b. stopping the first flow rate of the bore solution for a from a third surface and a fourth hole opposing the third
hole and extending into the reactor chamber from a
fourth surface;
surface:
surface:
surface:
surface:
third period; and
d, stopping the bore solution at a second f
- pporting and sealing the length of the first end of the d. stopping the second flow rate of the bore solution for hollow fiber in the first hole and the length of the $\frac{10}{4}$ a fourth period.
- capping the first and second ends of the hollow fiber with perature is decreased to a capping solution: a capping solution; of the first period.
- Exercise of the reactor module and fastening the reactor
module cover to the seal the reactor chamber;
module cover to the seal the reactor chamber;
fluidly connecting a bore solution to the first inlet;
fluidly connectin

fluidly connecting a shell solution to the second inlet; and flow rates are about 10 uniform flow rates are about 10 uniform growing the molecular sieving membrane that is uniform $_{20}$ individual hollow fiber.
and at least shout 0.5% free of defects on an inner hore 20 $\frac{1}{23}$. The method of claim 19, wherein the first period is and at least about 95% free of defects on an inner bore surface of the hollow fiber.

membrane is at least about 97% free of defects and wherein $\frac{10,00001}{20,00001}$ hours about 4 hours. any defects are less than about 20 nanometers in diameter, $\frac{10}{25}$ to about 4 nours.

24. The method of claim 19, further comprising the step

25 to about 4 nours .

25 of rinsing the membrane in solvents selected fro

membrane is at least about 97% free of defects and wherein consisting of 1-
any defects are less than about 10 nanometers in diameter, dejointed water.

11. The method of claim 8, wherein the bore solution providing a reactor cell device of claim 8, fluidically connecting a feed mixture to the first inlet; contains a limited $Zn+2$ concentration ranging from about fluidically connecting a feed mixture to the first inlet;
fluidically connecting a sweep gas to the second inlet;

12. The method of claim 8, wherein the bore solution collecting a separated mixture from the first outlet $\frac{1}{2}$ collecting a separated mixture from the second outlet. contains a limited $Zn+2$ concentration ranging from about 35 collecting permeate from the second outlet.
0.01 to about 0.03 mol/L $Zn+2$ in 1 octanol

13. The method of claim 8, wherein the bore solution is about 0.018 mol/L $Zn+2$ in 1-octanol.

14. The method of claim 8, wherein the shell solution
 $\frac{1}{27}$. The method of claim 25, wherein the feed mixture to contains an excess mum ligand concentration ranging from $\frac{27}{10}$. The method of claim 25, wherein the feed mixture to claim 0.5 to change 10 maple. 20 nother imideable (mum) in about 0.5 to about 10 mol/L 2-methyl imidazole (mum) in deionized water.

contains an excess mum ligand concentration ranging from the device comprises about 2 mol % to about 95 mol % n-butane and $\frac{1}{2}$ color 1.2 to about 1.6 mol/L.2 mol/mathematic comprises about 2 mol % to about 95 mol % about 1.2 to about 1.6 mol/L 2-methyl imidazole (mum) in $_{45}$ $_{\text{mixtures thereof.}}^{1-\text{outaire, about 2}}$ deionized water.

about 1.37 mol/L mum in deionized water. perature for the device is from about 3.37 mol/L mum in deionized water.

17. The method of claim 8, further comprising the step of or any value there between.
 $\frac{30}{25}$. The method of claim 25, wherein an operating presgently stirring the shell solution to prevent formation of $_{50}$ local concentration gradients.

18. The method of claim 14, wherein the shell solution is stirred at about 40 rpm to about 80 rpm.

- from the second surface;

from the second surface;
 $\frac{1}{2}$ period, wherein a first temperature is increased to a
 $\frac{1}{2}$ second temperature during a portion of the first period;
	-
	-
	-

hold and the first and second ends of the hollow fiber with **20**. The method of claim 19, wherein the second tem-
period the first and second ends of the hollow fiber with perature is decreased to a third temperature durin

fabricating the reactor module cover to fit on the upper $\frac{15}{10}$ and method of claim 19, wherein the first and second of claim 19 μ .

about 1 hour to about 3 hours, the second period is about 3 hours to about 4 hours, the third period is about 10 minutes 9. The method of claim 8, wherein the molecular sieving hours to about 4 hours, the third period is about 10 minutes on the third period is about 3 hours of the third period is about 3 hours of the third period is about 3

10. The method of claim 8, wherein the molecular sieving of rinsing the membrane in solvents selected from the group

any defects are less than about 10 nanometers in diameter, $\frac{25}{\text{A}}$ method of using a reactor cell device, comprising: length or width . $\frac{11}{\text{B}}$ is method of claim 8,

0.005 to about 0.1 mol/L Zn+2 in 1-octanol.
12 The mothod of claim 8, wherein the here solution collecting a separated mixture from the first outlet; and

0.01 to about 0.03 mol/L $\sum_{i=1}^{\infty}$ in 1-octanol.
13 The method of claim 8, wherein the hore solution is the device is selected from the group consisting of hydrogen/ hydrocarbons, hydrogen/propane, propylene/propane and butenes/butanes.

propane, propylene/propane and butenes/butanes.
28. The method of claim 26, wherein the feed mixture to

15. The method of claim 8, wherein the shell solution 28. The method of claim 26, wherein the feed mixture to the device comprises about 2 mol % to about 95 mol %

 $\frac{45 \text{ m}}{137 \text{ mol/L}}$ mum in deionized water . 29. The method of claim 25, wherein an operating tem $\frac{45 \text{ m}}{25 \text{ mol/L}}$ mum in deionized water

sure for the device is from about 1 bar to about 14 bar or any value there between.